

The influence of oxide buffer layers on the formation of catalytic nanoparticles and carbon nanotubes

© I.Kh. Khudaykulov, A.A. Rakhimov, A.A. Ismatov, M.M. Adilov

Institute of Ion-Plasma Technologies, Academy of Sciences of Uzbekistan, Tashkent, Uzbekistan
E-mail: i_khudaykulov@mail.ru

Received September 9, 2025

Revised October 18, 2025

Accepted October 30, 2025

The morphology of Ni nanoparticles synthesized on SiO₂ and TiO₂ buffer layers, as well as carbon nanotubes grown on them, was studied. Scanning electron microscopy and transmission electron microscopy analysis revealed that nanoparticles formed on the TiO₂ surface have increased size and low surface density, which leads to the formation of a limited number of carbon nanotube nuclei and reduced homogeneity. In contrast, nanoparticles on the SiO₂ surface have a smaller average diameter, a nearly spherical shape, a uniform distribution, and are characterized by high density. As a result, the synthesized carbon nanotubes are characterized by an ordered structure, a narrow diameter range (5–25 nm), and a high level of crystallinity. These results allow us to recommend the SiO₂ buffer layer as an effective catalyst platform.

Keywords: Carbon nanotubes, buffer layers, nanoparticles, CVD synthesis, morphology, transmission electron microscopy, scanning electron microscopy, crystallinity, amorphous carbon, catalytic particles.

DOI: 10.61011/TPL.2026.02.63050.20493

Large-scale materials and technology research has been conducted in recent years at the nanoscale. Owing to their high mechanical strength [1], unique electrical properties [2,3], high thermal conductivity [4,5], and lightweight [6], carbon nanotubes find application in various fields. They offer wide opportunities in the development of energy storage systems, integrated circuits [7,8], electrochemical capacitors [9], sensors, composites, and thermal interface materials [10–12].

The above applications require the synthesis of high-purity unidirectionally oriented carbon nanotubes (CNTs) with the desired parameters. Owing to its controllability and scalability, catalytic chemical vapor deposition (CVD) is regarded as the standard synthesis method at present. Initial formation of catalytic particles and prevention of their side reaction with the substrate are the prerequisites of efficient CVD. Catalysts are typically deposited in the form of a thin film and break up into nanoparticles upon heating. It was demonstrated that buffer layers provide an opportunity to control the shape and density of catalytic particles [13]. Metal buffer layers turned out to be inefficient [14], but the use of nitride and oxide buffers improves significantly the quality and yield of CNTs [15–17]. For example, significant differences were revealed when the influence of different buffer layers (Al, Al₂O₃, TiN, and TiO₂) on the Fe catalyst was examined [14]. In addition, it was found that the morphology of the buffer layer and the method of its formation affect directly the density and morphology of the catalyst and the formation of CNT arrays [18]. Thus, the choice of effective buffer layer and catalyst materials is crucial for high-quality CNT synthesis.

The present study is focused on the evaluation of efficiency of TiO₂ and SiO₂ buffer layers on single-crystal silicon in the formation of catalytic particles and the synthesis of carbon nanotubes.

An *n*-type Si(110) substrate with a diameter of 7.6 cm and a thickness of 376 μm was prepared for the synthesis of catalytic Ni particles. The substrate was cleaned with acetone and isopropyl alcohol, washed in an ultrasonic bath, and dried in Ar atmosphere.

The TiO₂ buffer layer was formed by atomic layer deposition (ALD, SENTECH Instruments GmbH) with titanium isopropoxide and water used as precursors. Nitrogen was used as the transport gas at a flow rate of 120 cm³/min. The pressure in the reactor was close to 60 Pa, and the substrate temperature was 200 °C. The deposition process consisted of 200 cycles of 8 s each, resulting in a film thickness of approximately 40 nm. The SiO₂ buffer layer was obtained by thermal oxidation at a temperature of 1000 °C for 2 h in an SNOL-type furnace in air. The approximate thickness of the formed layer was 55 nm. The nickel coating was deposited by electron-beam PVD (EB-PVD) at a pressure of approximately 10⁻⁶ Torr and a substrate temperature of 20 °C. The emission current was 50 mA, the accelerating voltage was 10 kV, the distance between the substrate and the target was 25 cm, and the deposition time was 40 s. The film growth rate was determined based on the spectroscopic ellipsometry data and was found to be approximately equal to 0.3 nm/s. Prior to the synthesis of carbon nanotubes, the nickel catalyst was activated in an H₂ : Ar = 1 : 3 mixture of hydrogen and argon for 10 min. The activation time was chosen with account for the initial Ni layer thickness and the heating temperature, which ensured uniform formation of

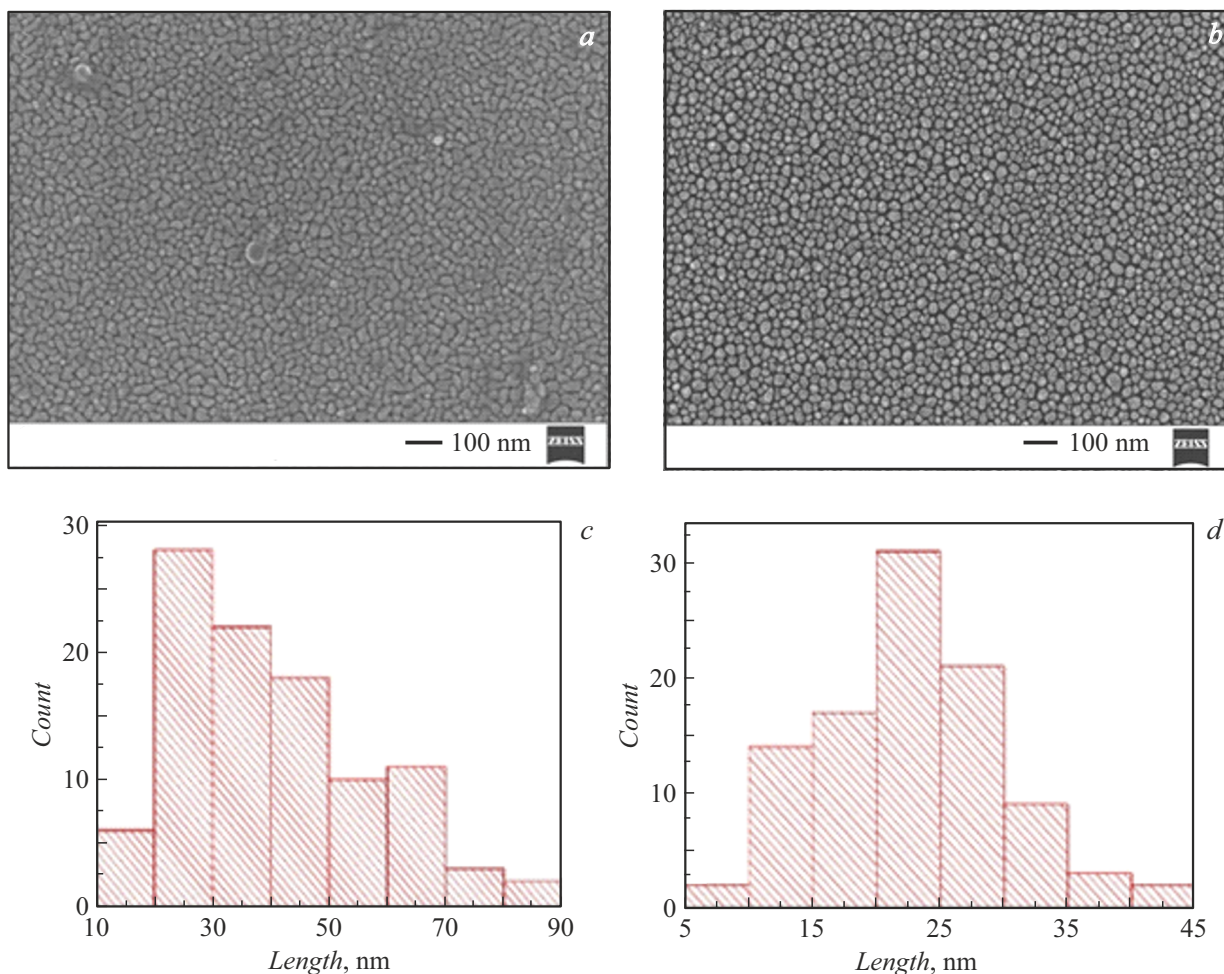


Figure 1. SEM images of Ni catalyst particles formed on the TiO₂/Si (a) and SiO₂/Si (b) samples and statistical histograms of the nanostructure distribution (c,d).

catalytic nanoparticles. The samples were heated in a CVD chamber at a temperature of 700 °C in an atmosphere of H₂ and Ar; after that, CH₄ was supplied, and CNT synthesis was carried out for 30 min at Ar : H₂ : CH₄ = 5 : 1 : 2. The thickness of buffer layers measured by spectroscopic ellipsometry (SENTECH SER850) was 40 nm for TiO₂ and 55 nm for SiO₂. According to the results of earlier studies [19], the TiO₂ layer is characterized by an anatase phase with a nanocrystalline structure, while SiO₂ is amorphous. Scanning electron microscopy (SEM; ZEISS Sigma 500), transmission electron microscopy (TEM; HRTEM JEM 3010 URP-JEOL), and Raman spectroscopy (Renishaw RM 100 with a laser wavelength of 785 nm and a laser power of 1 mW) were used for analysis. Since local destruction of the sample surface was observed at higher powers, measurements were carried out at the minimum radiation power.

The morphology of the Ni thin film was investigated by SEM. Figures 1, a, b present the images of Ni particles formed after heating on the TiO₂ and SiO₂ buffer layers. On the TiO₂ surface, they assume the form of islands of varying sizes and irregular shape; on SiO₂, they are near-spherical in shape and have a smaller average diameter and a more

uniform distribution. This suggests that the SiO₂ buffer layer is more efficient in forming high-density catalytic particles and distributing them uniformly over the surface. The thermal decomposition of the thin Ni film is associated with minimization of its surface energy, and this process is effected through surface diffusion [19]. SEM images were analyzed using ImageJ. On the TiO₂/Si surface, the average particle size was 40.8 nm at a density of 853 μm⁻²; on the SiO₂/Si surface, the average size and the density were 22.8 nm and 1090 μm⁻², respectively. The histogram did also reveal that particles on the SiO₂/Si surface are significantly smaller and are distributed within a narrower range. The Raman spectra shown in Figs. 2, a, b are indicative of the formation of multi-walled CNTs in both cases. The RBM peak was not observed; the G peak was observed at 1591 cm⁻¹ (TiO₂) and 1574 cm⁻¹ (SiO₂), while the D peak was identified at 1357 and 1340 cm⁻¹, respectively. Second-order peak G' was also detected in both samples. The D/G ratio was 0.79 for TiO₂ and 0.53 for SiO₂, indicating a higher purity of CNTs synthesized on the SiO₂ buffer layer [20].

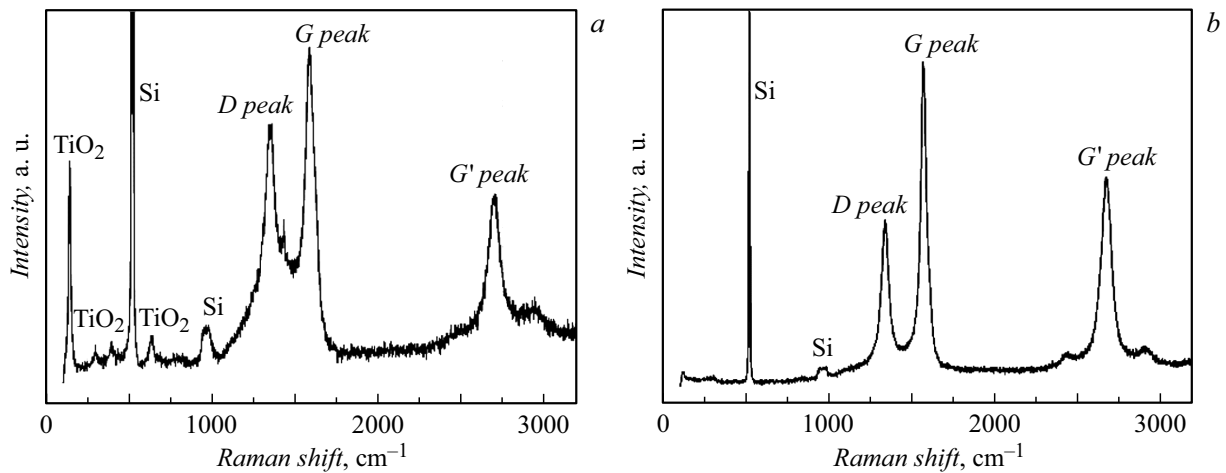


Figure 2. Raman spectra of CNTs grown on the Ni/TiO₂/Si (a) and Ni/SiO₂/Si (b samples).

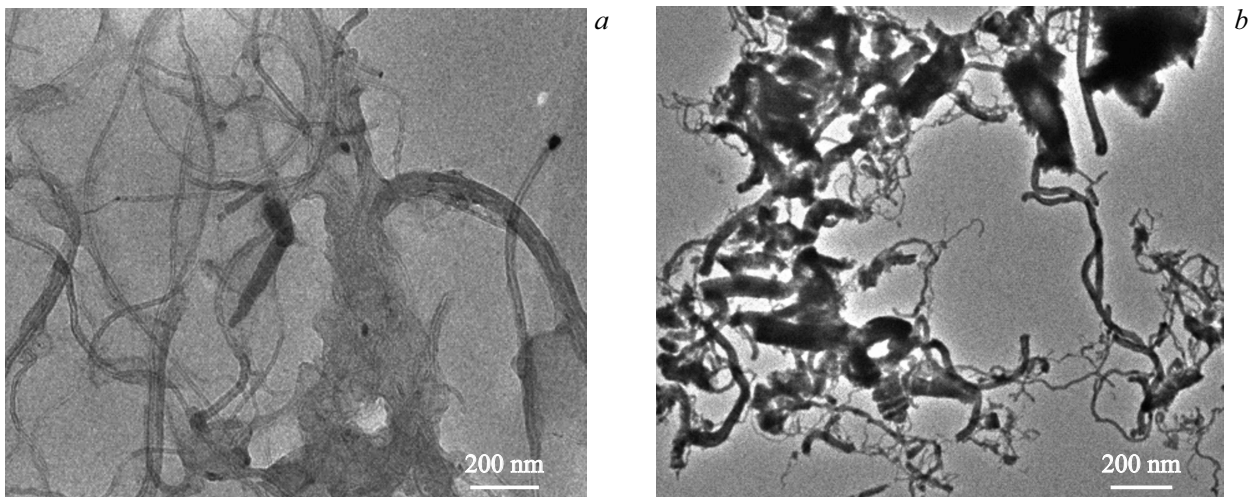


Figure 3. TEM images of multi-walled carbon nanotubes synthesized on the Ni/SiO₂/Si (a) and Ni/TiO₂/Si (b samples).

The TEM images shown in Fig. 3 reveal that CNTs on the SiO₂ surface have a multi-walled structure with a diameter of 5–25 nm and a length of several tens of micrometers. They are randomly oriented, but have a smooth surface, and Ni particles were found at their ends. A small amount of CNTs with a diameter of 5–40 nm did form (in most cases, together with amorphous carbon) in the sample synthesized on the TiO₂ surface. This is associated with the enlargement of catalytic particles, which leads to suppression of their activity and premature termination of synthesis.

Thus, Ni nanoparticles formed on SiO₂ and TiO₂ buffer layers differ significantly in morphology and density, which has a direct influence on the quality and structure of carbon nanotubes synthesized based on them. Owing to the formation of large nanoparticles with low density, the growth of CNTs with an admixture of amorphous carbon and a less ordered structure was observed on the TiO₂ surface. At the same time, CNTs with a narrow range of diameters and high degrees of crystallinity and purity were obtained on the SiO₂ surface, where nanoparticles with a

smaller average diameter, a near-spherical shape, and a high density were formed. The $I_D/I_G = 0.53$ intensity ratio in the Raman spectra verifies the high quality of synthesized CNTs. Therefore, SiO₂ buffer layers may be regarded as an efficient support for controlled growth of high-quality nanocarbon structures.

Conflict of interest

The authors declare that they have no conflict of interest.

References

- [1] R.S. Ruoff, D. Qian, W.K. Liu, *Compt. Rend. Phys.*, **4** (9), 993 (2003). DOI: 10.1016/j.crhy.2003.08.001
- [2] X.F. Sánchez-Romate, A. Jiménez-Suárez, A. Ureña, in *Handbook of carbon nanotubes* (Springer, Cham, 2022), p. 213–247.
- [3] C.J. Barnett, C.E. Gowenlock, K. Welsby, A. Orbaek White, A.R. Barron, *Nano Lett.*, **18** (2), 695 (2018). DOI: 10.1021/acs.nanolett.7b03390

- [4] S. Berber, Y.K. Kwon, D. Tománek, *Phys. Rev. Lett.*, **84** (20), 4613 (2000). DOI: 10.1103/PhysRevLett.84.4613
- [5] J. Hone, M. Whitney, C. Piskoti, A. Zettl, *Phys. Rev. B*, **59** (4), R2514 (1999). DOI: 10.1103/PhysRevB.59.R2514
- [6] A. Wang, Z. Zhang, Y. Liu, Z. Li, J. Leng, *Carbon*, **225**, 119105 (2024). DOI: 10.1016/j.carbon.2024.119105
- [7] M. Horibe, M. Nihei, D. Kondo, A. Kawabata, Y. Awano, *Jpn. J. Appl. Phys.*, **44** (7R), 5309 (2005). DOI: 10.1143/JJAP.44.5309
- [8] A.T. Lawal, *Carbon Trends*, **19**, 100470 (2025). DOI: 10.1016/j.cartre.2025.100470
- [9] G.G. Bizuneh, A.M. Adam, J. Ma, *Battery Energy*, **2** (1), 20220021 (2023). DOI: 10.1002/bte2.20220021
- [10] L. Ci, J. Suhr, V. Pushparaj, X. Zhang, P.M. Ajayan, *Nano Lett.*, **8** (9), 2762 (2008). DOI: 10.1021/nl8012715
- [11] H.M. Dewey, A. Lamb, J. Budhathoki-Uprety, *Nanoscale*, **16** (35), 16344 (2024). DOI: 10.1039/d4nr01892c
- [12] A. Nylander, J. Hansson, M. Kabiri Samani, C. Chandra Darmawan, A. Borta Boyon, L. Divay, J. Liu, *Energies*, **12** (11), 2080 (2019). DOI: 10.3390/en12112080
- [13] Y. Wang, B. Li, P.S. Ho, Z. Yao, L. Shi, *Appl. Phys. Lett.*, **89** (18), 183113 (2006). DOI: 10.1063/1.2382735
- [14] T. de los Arcos, M.G. Garnier, P. Oelhafen, D. Mathys, J.W. Seo, C. Domingo, J.V. García-Ramos, S. Sánchez-Cortés, *Carbon*, **42** (1), 187 (2004). DOI: 10.1016/j.carbon.2003.10.020
- [15] Y.J. Jung, B. Wei, R. Vajtai, P.M. Ajayan, Y. Homma, K. Prabhakaran, T. Ogino, *Nano Lett.*, **30** (4), 561 (2003). DOI: 10.1021/nl034075n
- [16] N. Nagaraju, A. Fonseca, Z. Konya, J.B. Nagy, *J. Mol. Catal. A*, **181** (1-2), 57 (2002). DOI: 10.1016/S1381-1169(01)00375-2
- [17] G.B. Zheng, H. Sano, Y. Uchiyama, *Mater. Sci. Forum*, **544-545**, 773 (2007). DOI: 10.4028/www.scientific.net/MSF.544-545.773
- [18] Y. Wang, Z. Luo, B. Li, P.S. Ho, Z. Yao, L. Shi, E.N. Bryan, R.J. Nemanich, *J. Appl. Phys.*, **101** (12), 124310 (2007). DOI: 10.1063/1.2749412
- [19] T.K. Turdaliev, K.K. Zokhidov, F.I. Abdurakhmanov, A.A. Rakhimov, K.B. Ashurov, *Phys. Solid State*, **66** (7), 1158 (2024). DOI: 10.61011/PSS.2024.07.58992.115.
- [20] R. Saito, A. Grüneis, Ge.G. Samsonidze, V.W. Brar, G. Dresselhaus, M.S. Dresselhaus, A. Jorio, L.G. Cançado, C. Fantini, M.A. Pimenta, A.G. Souza Filho, *New J. Phys.*, **5** (1), 157 (2003). DOI: 10.1088/1367-2630/5/1/157

Translated by D.Safin