

NMR of monoclinic silver sulfide α -Ag₂S

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¹⁰⁹Ag-NMR spectrometry was used to examine the structure of monoclinic α -Ag₂S powder. It was found that the ¹⁰⁹Ag-NMR spectrum of monoclinic α -Ag₂S had a form of a single narrow line, whose width slightly varied with temperature within 85–295 K. At a temperature below 200 K, a considerable growth of the isotropic component of shift tensor K_{iso} of ¹⁰⁹Ag nuclei is observed in the Ag₂S powder. Abnormally short time T_2 of ¹⁰⁹Ag spin-spin relaxation was detected. Simulation shows that, besides the acanthite described in the literature, other Ag₂S phases, that are derived from high-temperature argentite, β -Ag₂S, and have different structures, may be formed in silver sulfide as the temperature decreases. It is shown that the band structure of all predicted model Ag₂S phases has a band gap of 0.6–1.5 eV that is indicative of semiconductor properties of the phases.

Keywords: silver sulfide, ¹⁰⁹Ag isotope, isotropic shift, spin-spin relaxation time, predicted low-temperature Ag₂S phases.

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1. Introduction

Nanostructured semiconductor sulfides are very important for various applications. Among them, large focus is made on nanostructured Ag₂S [1–3] that is a semiconductor at a temperature below 450 K and a superionic conductor at a temperature above 452 K.

According to the phase diagram of the Ag-S system [4,5], Ag₂S has three main polymorphous modifications. Low-temperature semiconductor α -Ag₂S phase with a monoclinic structure (acanthite) exists at a temperature below $\sim$ 450 K. Cubic β -Ag₂S phase (argentite) in equilibrium conditions exists in the temperature range of 452–859 K, has an body-centered (BCC) sublattice of S atoms and superionic conductivity. High-temperature cubic γ -Ag₂S phase with face-centered cubic sublattice of S atoms is stable at a temperature from $\sim$ 860 K to the melting temperature.

Potential use of nanostructured silver sulfide is most promising in micro/nanoelectronics, where Ag₂S/Ag heteronanostructures are used in nonvolatile memory and resistive switches. Their action is based on the reduction of Ag⁺ to metallic Ag atoms, transformation of acanthite, α -Ag₂S, to argentite, β -Ag₂S, and appearance of conducting channel consisting of Ag and β -Ag₂S [6,7]. Due to the features of a band structure, Ag₂S/Ag hybrid heteronanostructures displayed high efficiency in heterogeneous catalysis and photocatalysis [8,9].

Ag₂S is used as an optical and electronic component (photovoltaic cells, photoconductors, biosensors and biomarkers, IR detectors) [10] and as an superionic conductor [11–13]. All types of nanostructured silver sulfide

have considerable antibacterial activity. Ag₂S is used in photoswitches and oxygen sensors.

Ag₂S quantum dots demonstrate high photoluminescence and outstanding photostability. Therefore, Ag₂S quantum dots smaller than 3–4 nm may be used as luminophores or biomarkers for infrared radiation.

2. Results and discussion

Silver sulfide powders were synthesized by chemical deposition from AgNO₃, Na₂S and Na₃C₆H₅O₇ solutions in water. Solubility product K_{sp} of silver sulfide is small and is $K_{sp} = 6.3 \cdot 10^{-50}$ at 298 K [3], therefore, silver sulfide deposition occurs almost immediately. Procedure for synthesis of Ag₂S powders with the desired mean size of nanoparticles was previously described in detail in [3,14].

Synthesized silver sulfide powders were examined using the Shimadzu XRD-7000 diffractometer in CuK α_1 - radiation in the angle interval $2\theta = 20$ – 95° in $\Delta(2\theta) = 0.02^\circ$ steps and with good statistics. Determination of lattice parameters and final clarification of the synthesized silver sulfide powder structure were performed in X'Pert HighScore Plus software package [15]. Mean size D of particles (more precisely — coherent scattering regions (CSR)) in the synthesized microcrystalline silver sulfide powders was evaluated by X-ray diffraction according to diffraction reflection broadening $\beta(2\theta)$ using the dependence of reduced $\beta^*(2\theta) = [\beta(2\theta) \cos \theta]/\lambda$ on the scattering vector $s = (2 \sin \theta)/\lambda$.

X-ray pattern of the synthesized monoclinic α -Ag₂S powder is shown in Figure 1. Many diffraction reflections of

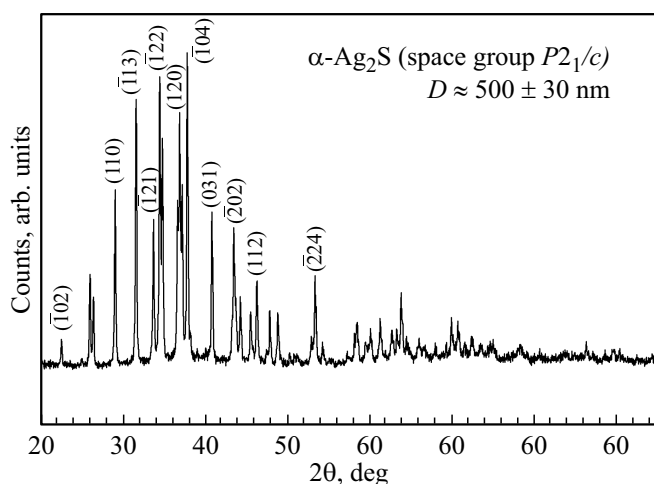


Figure 1. X-ray pattern of the microcrystalline monoclinic (space group $P2_1/c$) (a) α -Ag₂S powder at 295 K with particle size D equal to ~ 500 – 30 nm.

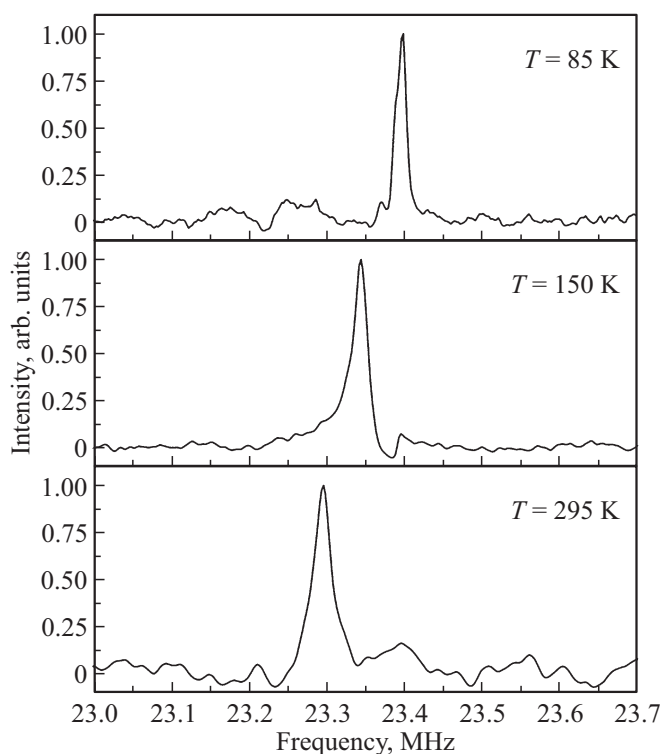


Figure 2. Typical ^{109}Ag -NMR spectra in the microcrystalline Ag₂S powder recorded at various temperatures in the range of 85–295 K.

the powder are overlapped. Quantitative clarification of the silver sulfide powder X-ray pattern and comparison with data of [14] showed that the observed set of diffraction reflections corresponded to monoclinic (space group $P2_1/c$) acanthite, α -Ag₂S. Evaluation of the CSR powder mean size conducted using nonoverlapping diffraction reflections showed that the size was approx. 500 nm, i.e. this was a microcrystalline powder.

^{109}Ag -NMR spectra in the microcrystalline Ag₂S powder were recorded on the AVANCE III 500 (Bruker) NMR spectrometer using a special low-temperature cell in a continuous-flow cryostat in the temperature range from 85 K to 295 K in the $H_0 = 11.7467$ T external magnetic field. NMR spectra were recorded using a standard double-pulse spin-echo sequence $\tau - t_{\text{del}} - \tau - t_{\text{del}} - \text{echo}$. Duration of the first pulse was set to $\tau = 2 \mu\text{s}$. RF transmitter power was 400 W. NMR signal was recorded with delay between pulses $t_{\text{del}} = 20 \mu\text{s}$. Spin-spin relaxation time was measured according to a standard procedure using the same sequence. In all cases, nuclear magnetization recovery was adequately described by a single exponential function [16].

To date, there is no literature data concerning ^{109}Ag -NMR spectral measurements of silver sulfide. ^1H - and ^{13}C -NMR spectral measurements of platelet silver thiolates (Ag-MPA) are shown in [17]. Thiolate complexes are used as capping agents in the synthesis of monodispersed silver, gold and other metal clusters.

Figure 2 shows typical ^{109}Ag -NMR spectra in microcrystalline Ag₂S powder recorded at a temperature from 85 K to 295 K in the 117.467 kOe external magnetic field. There are two naturally-occurring silver isotopes: ^{107}Ag (natural content 51.8%) and ^{109}Ag (natural content 48.2%), both isotopes have the nuclear spin $I = 1/2$. ^{109}Ag nuclei were chosen for measurements because their higher gyromagnetic ratio provides a better signal-to-noise ratio. ^{109}Ag -NMR spectrum of a microcrystalline sample (Figure 2) is a single narrow line with FWHM ~ 20 kHz.

For the monoclinic crystalline structure of Ag₂S with equivalent positions of Ag atoms, particularly a single line shall be expected because nuclei with spin 1/2 has no quadrupole moment. Therefore only a central transition may be observed, i.e. the central part of spectrum associated with transition ($m = 1/2 \leftrightarrow -1/2$), where m is the magnetic quantum number [18,19]. Line width slightly varies with temperature in the range of 85–295 K. Analysis of ^{109}Ag -NMR spectra in the microcrystalline Ag₂S powder determined the shift tensor components. It was found that the shift was isotropic (within the measurement accuracy): $K_{\text{iso}} = 0.12(2)\%$ ($K_x = K_y = K_z$).

Temperature dependence of the isotropic component of shift tensor K_{iso} of ^{109}Ag nuclei in the microcrystalline Ag₂S powder is shown in Figure 3. Figure 3 shows that the shift of the NMR line (K_{iso}) varies slightly as the temperature decreases from 295 K to 200 K, whereas a significant shift growth is observed at a temperature below 200 K. Such behavior is not quite usual for a semiconductor compound without magnetic ions. The reason for the observed effect may be in any structural phase transition at ~ 200 K in silver sulfide.

Temperature dependence of ^{109}Ag spin-spin relaxation time in Ag₂S is shown in Figure 4. Abnormally low T_2 is observed at 250 K (7 mks); it is even less at room temperature. Such times may be associated with fast dynamic processes near the Ag₂S ions. In particular, this

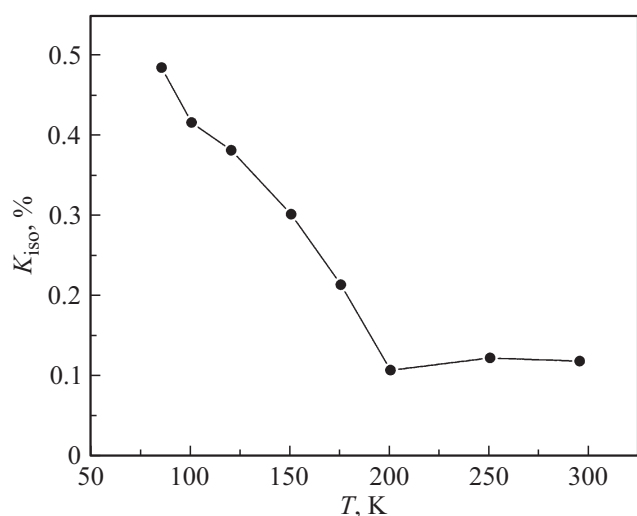


Figure 3. Temperature dependence of the isotropic component of shift tensor K_{iso} of ^{109}Ag nuclei in the microcrystalline Ag_2S powder.

may be associated with ionic mobility of silver that is typical of BCC argentite, $\beta\text{-Ag}_2\text{S}$.

Generally, the review of the ^{109}Ag -NMR spectra of the microcrystalline Ag_2S powder detected an abnormally low spin-spin relaxation time T_2 and significant temperature dependence of shift K_{iso} , which is not typical of the monoclinic semiconductor acanthite, $\alpha\text{-Ag}_2\text{S}$. NMR is used for experimental examination of short-range order features, therefore the considerable growth of NMR line shift (K_{iso}) observed at a temperature below 200 K may result from a change of symmetry of the studied silver sulfide. In particular, the identified large growth of K_{iso} of the NMR line at a temperature below 200 K (Figure 3) indirectly suggests that there is a low-temperature Ag_2S phase at $T < 200$ K. Potential existence of low-temperature silver sulfide phases with cubic, tetragonal, orthorhombic, trigonal, monoclinic and triclinic symmetry was addressed previously in [20]. Possible model Ag_2S phases in [20] were sought using USPEX (Universal Structure Predictor: Evolutionary Xtallography) [21] described in detail in [22].

According to [2,23], the $\alpha\text{-Ag}_2\text{S}$ structure results from the distortion of the BCC sublattice of S atoms in the $\beta\text{-Ag}_2\text{S}$ structure. Argentite — acanthite transformation is followed by the distortion of BCC sublattice of S atoms to a monoclinic sublattice and the shift of S and Ag atoms [3]. Transformation of the BCC argentite into the low-temperature monoclinic acanthite may be generally treated as ordering in the BCC sublattice of S atoms [24,25]. An argentite ordering option with formation of the monoclinic (space group $P2_1$) Ag_2S phase was proposed in [26]. In [20], it is shown that, besides the acanthite, other Ag_2S phases, that are derived from high-temperature argentite, $\beta\text{-Ag}_2\text{S}$, and have different structures, may be formed in silver sulfide as the temperature decreases.

Calculation of the enthalpy of formation of all given model Ag_2S phases [20] showed that the monoclinic $\alpha\text{-Ag}_2\text{S}$ structure (Figure 5, *a*) described in the literature [4,23,27] was not the only possible and the most energy-favorable low-temperature silver sulfide phase; its enthalpy of formation ΔH_f is equal to -0.033 eV per formula unit. The calculations gave the monoclinic (space group $P2_1/c$)

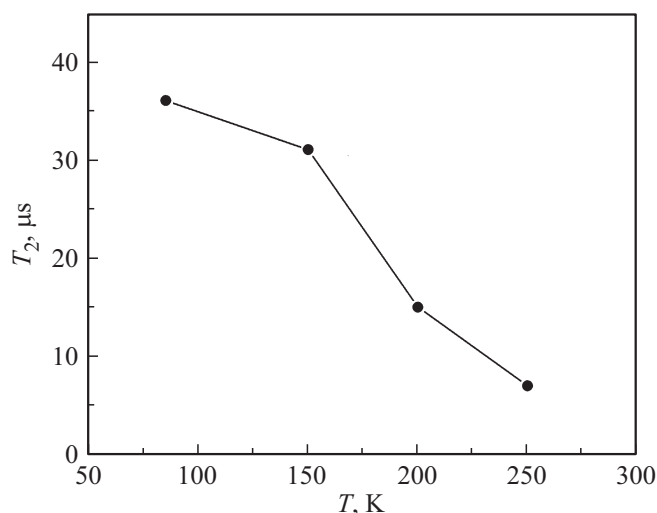


Figure 4. Temperature dependence of ^{109}Ag spin-spin relaxation time T_2 in microcrystalline Ag_2S powder.

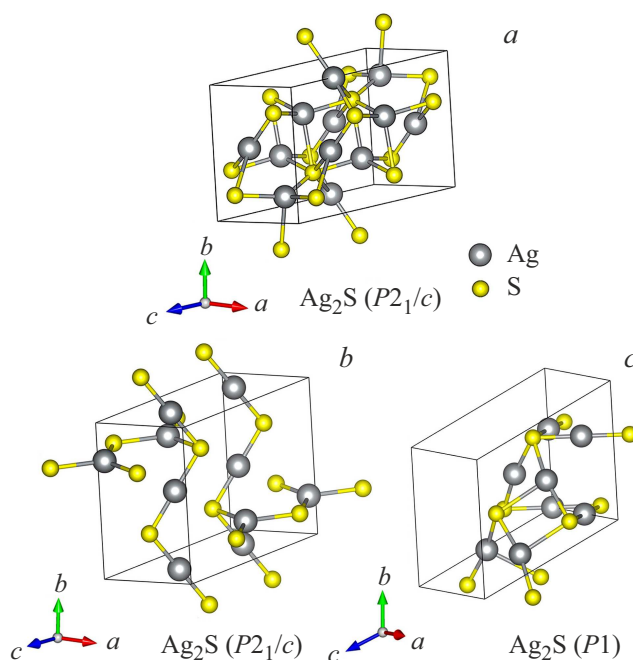


Figure 5. Lattice cell (*a*) of monoclinic (space group $P2_1/c$) $\alpha\text{-Ag}_2\text{S}$ according to [23,24] compared with the predicted model lattice cells: *b* — monoclinic (space group $P2_1/c$) Ag_2S phase with the enthalpy of formation $\Delta H_f = -0.219$ eV/formula units, which is the lowest among the predicted monoclinic structures, and *c* — triclinic (space group $P1$) Ag_2S phase with the lowest enthalpy of formation $\Delta H_f = -0.223$ eV/formula units.

Ag₂S phase (Figure 5, *b*) with a lower enthalpy of formation $\Delta H_f = -0.219$ eV per formula unit compared with the acanthite described in the literature and the triclinic (space group *P*1) Ag₂S phase (Figure 5, *c*) with the lowest enthalpy of formation $\Delta H_f = -0.223$ eV per formula unit in the ground state at $T = 0$ K and $P = 0$ Pa. Figure 5 shows the difference in the immediate S atom environment of Ag atoms in the monoclinic acanthite (Figure 5, *a*) and two model silver sulfide phases (Figure 5, *b* and *c*). Actually, according to the calculation [20], the interatomic distances Ag-S in these two model phases are slightly different from the interatomic distances Ag-S in monoclinic α -Ag₂S [3,23], which confirms some short-range order difference in them from the monoclinic acanthite and agrees with the obtained NMR measurements.

The band structure of all predicted model Ag₂S phases has a band gap from ~ 0.6 eV to ~ 1.5 eV that is indicative of semiconductor properties of the phases. Band gap E_g of triclinic (space group *P*1) Ag₂S is equal to 1.16 eV. Thus, theoretical calculations [20] of low-temperature silver sulfide phases confirm possible existence of the low-temperature Ag₂S phase at $T < 200$ K.

3. Conclusion

The ¹⁰⁹Ag-NMR method was first used to study the monoclinic α -Ag₂S powder. Measurements showed that the ¹⁰⁹Ag-NMR spectrum of monoclinic α -Ag₂S had a form of a single narrow line, whose width slightly varied at a temperature from 85 K to 295 K. Isotropic shift K_{iso} of the NMR line slightly varies as the temperature decreases from 295 K to 200 K, but at a temperature below 200 K a significant growth of the isotropic component of shift tensor K_{iso} of ¹⁰⁹Ag nuclei is observed in the Ag₂S powder. It is suggested that the observed effect may be associated with the structural phase transitions in silver sulfide at ~ 200 K.

Possible existence of low-temperature silver sulfide phases derived from high-temperature β -Ag₂S was addressed. The model calculations gave the monoclinic (space group *P*2₁/*c*) Ag₂S phase with a lower enthalpy of formation $\Delta H_f = -0.219$ eV per formula unit compared with the acanthite described in the literature and the triclinic (space group *P*1) Ag₂S phase with the lowest enthalpy of formation $\Delta H_f = -0.223$ eV per formula unit in the ground state at $T = 0$ K and $P = 0$ Pa. A band gap present in the band structure of all predicted model Ag₂S phases is indicative of their semiconductor properties.

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Conflict of interest

The authors declare that they have no conflict of interest.

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