Nature of the local environment of germany atoms in amorphous films $(GeTe)_x(Sb_2Te_3)$

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Using Messbauer spectroscopy on ¹¹⁹Sn impurity atoms, it was demonstrated that in amorphous (GeTe)_x(Sb₂Te₃) films (where x = 0.5, 1, 2, 3) tin atoms replace atoms of tetravalent germanium, which forms a tetrahedral system of chemical bonds with atoms in its local environment. In crystalline films, tin replaces divalent six-coordinated germanium in positions 4*b* of the NaCl-type crystal lattice.

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Phase memory materials based on chalcogenide alloys are used to store and encode data in non-volatile memory devices. The prerequisite for this is a noticeable difference in conductivity, as well as in the reflectivity of their crystalline and amorphous phases [1]. It is believed that alloys with compositions lying on the GeTe–Sb₂Te₃ pseudobinary line are the most convenient materials for creating rewritable optical storage devices [{]2}. They have short crystallization times, ideal reversibility of transitions between amorphous and crystalline states, and high thermal stability. Obviously, to increase the recording density and to miniaturize memory devices, it is necessary to understand the details of the materials microstructure with a phase transition in both the crystalline and amorphous states.

At present, the crystal structures of $(GeTe)_x(Sb_2Te_3)$ alloys (where x = 0.5, 1, 2, 3) are studies in details [3–6]. But, contradictions in the interpretation of experimental results obtained in the study of amorphous films Ge₂Sb₂Te₅ by spectroscopy method of (extended X-ray absorption fine structure, EXAFS) [7–10]. This indicates the need to use other experimental methods more sensitive to minor changes in the local structure of the electron shells of atoms during the transition from amorphous to crystalline state. In particular, it was shown that Messbauer spectroscopy is an effective tool for detecting changes in the local environment of atoms and their electronic structure during amorphization of compounds of type Ge₂Sb₂Te₅ [11]. This paper relates to studying the nature of the local environment of germanium atoms in crystalline and amorphous films $(GeTe)_x(Sb_2Te_3)$ (where x = 0.5, 1, 2, 3) by absorption Messbauer spectroscopy on the isotope ¹¹⁹Sn.

X-ray amorphous films a-Ge₃Sb₂Te₆, a-Ge₂Sb₂Te₅, a-GeSb₂Te₄, a-GeSb₄Te₇ (denoted as a-GeSbTe) and a-Ge_{2.95}Sn_{0.05}Sb₂Te₆, a-Ge_{1.95}Sn_{0.05}Sb₂Te₅, *a*-Ge_{0.95}Sn_{0.05}Sb₂Te₄ and *a*-Ge_{0.95}Sn_{0.05}Sb₄Te₇ (let's denote them as *a*-Ge(Sn)SbTe) were obtained by magnetron sputtering of polycrystalline targets of the same composition at direct current in a nitrogen atmosphere on aluminum foil substrates. Then the films were annealed at temperatures of 150–200°C to obtain *c*-Ge₃Sb₂Te₆, *c*-Ge₂Sb₂Te₅, *c*-GeSb₂Te₄, *c*-GeSb₄Te₇ (denoted as *c*-GeSbTe) and *c*-Ge_{2.95}Sn_{0.05}Sb₂Te₆, *c*-Ge_{1.95}Sn_{0.05}Sb₂Te₅, *c*-Ge(Sn)SbTe). The described compositions are given for sputtered targets. The thickness of the films was determined using an atomic force microscope and ranged from 40 to 120 μ m. For the synthesis of tin-containing alloys, the ¹¹⁹Sn isotope of 96

Germanium doped with tin was obtained by sputtering a film of metallic tin (enriched to 96

The Messbauer spectra of ¹¹⁹Sn were recorded on a SM 4201 TerLab spectrometer at 80 K with a Ca^{119mm}SnO₃ source. The instrumental spectral linewidth (G_{app}) for the ¹¹⁹Sn isotope was 0.79(2) mm/s. Isomeric shifts δ of ¹¹⁹Sn spectra are given relative to the CaSnO₃ absorber. The Messbauer spectra of ¹¹⁹Sn atoms in the *a*-Ge(Sn)SbTe films are shown in Figs 1 and 2. In interpreting the Messbauer spectroscopy data, it was assumed that the impurity tin atoms isovalently substitute for germanium atoms in amorphous and crystalline films.

The Messbauer spectra of all *a*-Ge(Sn)Sb films are single broadened lines ($G \sim 1.33-1.36$ mm/s) with isomeric shifts $\delta \sim 2.04-2.07$ mm/s. These shifts are close to the isomeric shifts of the spectra of impurity tin atoms in crystalline germanium (Fig. 1), as well as to the isomeric shift of the α -Sn gray tin spectrum. The isomeric shifts of these two spectra ($\delta = 1.79$ mm/s for Ge and $\delta = 2.10$ mm/s for α -Sn) form the region of isomeric shifts



Figure 1. Messbauer spectra ¹¹⁹Sn of *a*-Ge_{2.95}Sn_{0.05}Sb₂Te₆ (*a*), *a*-Ge_{1.95}Sn_{0.05}Sb₂Te₅ (*b*), a-Ge_{0.95}Sn_{0.05}Sb₂Te₄ (*c*), *a*-Ge_{0.95}Sn_{0.05}Sb₄Te₇ (*d*) and crystalline germanium (*e*).

of tetravalent tin compounds with tetrahedral sp^3 -system of chemical bonds. Hence, impurity tin atoms in the structure of a-Ge(Sn)Sb films isovalently replace the fourvalent germanium atoms, which form a tetrahedral chemical bond, i.e., local coordination number of germanium atoms in amorphous films is four).

Note that if in the *a*-Ge(Sn)SbTe films in the local environment of germanium atoms (and tin atoms replacing them) there are only germanium atoms, then the isomeric shift of the spectra ¹¹⁹ The Sn of the films should be ~ 1.8 mm/s (as in the spectrum of impurity atoms ¹¹⁹Sn in germanium). If in the *a*-Ge(Sn)SbTe films in the local environment of germanium atoms (and tin atoms replacing them) there are only tellurium atoms, then the isomeric shift of the spectra ¹¹⁹ The Sn of the films should be ~ 2.1 mm/s (as in the ¹¹⁹Sn spectrum of Ge_{1.45}Sn_{0.05}Te_{8.5} glassy alloys [12]). Based on this, it should be concluded that the ¹¹⁹Sn Messbauer spectra of the *a*-Ge(Sn)SbTe films correspond to germanium atoms (and tin atoms

replacing them), in the local environment of which there are germanium atoms, and predominantly tellurium atoms.

Kolobov et al. [7] to interpret the EXAFS data obtained in the study of the order-disorder transition in Ge₂Sb₂Te₅ suggested the model of "umbrella flip". According to this model, film amorphization is accompanied by a jump of the Ge atom from the octahedral position occupied in the crystalline film to the tetrahedral position surrounded by four tellurium atoms, and Ge-Ge bonds were not found. However, Baker et al. [8,9], also using EXAFS data to study films a-Ge₂Sb₂Te₅, propose a model according to which germanium atoms form the structural units Te₃Ge-GeTe₃, and the model is based on the predominant formation of bonds in amorphous films Ge-Ge. Finally, the structure of a-Ge₂Sb₂Te₅ and a-GeSb₂Te₄ films was examined by the authors [10] by the EXAFS method also. It was shown that Ge-Ge and Ge-Sb bonds are present in crystalline and amorphous films, while Te-Te and Sb-Sb bonds were not



Figure 2. Messbauer spectra ¹¹⁹Sn of c-Ge_{2.95}Sn_{0.05}Sb₂Te films ₆ (*a*), c-Ge_{1.95}Sn_{0.05}Sb₂Te₅ (*b*), c-Ge_{0.95}Sn_{0.05}Sb₂Te₄ (*c*), *a*-Ge_{0.95}Sn_{0.05}Sb₄Te₇ (*d*) and SnTe (*e*) compounds.

found. All atoms satisfy the formal valence requirements, and the germanium atoms have quadruple coordination.

The Messbauer spectroscopy data obtained allow us to conclude that tetravalent germanium atoms form a tetrahedral sp^3 system of chemical bonds in the structural grid of the amorphous matrix (local coordination number of germanium atoms is four) and have mainly tellurium atoms in their closest environment (although germanium atoms are not excluded).

A characteristic feature of the spectra of impurity tin atoms in amorphous films is their broadening, and two reasons for this broadening can be proposed. The first reason is that possible distortions of the angles between the bonds of tin atoms with tellurium atoms in its nearest environment lead to spectrum broadening due to the unresolved quadrupole splitting QS ~ 0.55 mm/s, which indicates a significant distortion tetrahedral bond angles. The second reason for the spectra broadening is fluctuations in the distances from tin atoms to tellurium atoms, while maintaining the tetrahedral system of chemical This leads to spectrum broadening due to the bonds. heterogeneous isomeric shift. Using the calibration of the isomeric shifts of the ¹¹⁹Sn Messbauer spectra given in [13], we found that the electronic configurations of tin lie within $5s^{0.92}p^{2.76}$ to $5s^{1.15}p^{3.45}$, and the most probable configuration $5s^{1.01}p^{3.03}$ corresponds to the atomic configuration when four atoms of tellurium are at equal distances in the nearest environment of tin atoms.

The Messbauer spectra ¹¹⁹Sn of all *c*-Ge(Sn)Sb films are single broadened lines ($G \sim 1.30-1.35$ mm/s) with isomeric shifts within $\delta \sim 3.49-3.51$ mm/s. So, spectra of *c*-Ge(Sn)SbTe films have isomeric shifts typical of the spectra of ¹¹⁹Sn ion compounds of divalent tin. For understanding, Fig. 2 shows the spectrum of the divalent tin compound SnTe, for which $\delta = 3.54$ mm/s.

c-GeSbTe compounds have three crystalline phases: a metastable vacancy-disordered cubic phase (can be obtained by heating an amorphous sample above the crystallization temperature of 150°C), a vacancy-ordered cubic phase (can be obtained by annealing amorphous samples at 300°C) and the hexagonal phase (obtained by prolonged heating of an amorphous sample at a temperature above 300°C) [3,4]. The general structure of these compounds consists of rock salt-type construction blocks with alternating cationic (GeSb) and anionic (Te) layers [3,4].

The Messbauer spectroscopy data on ¹¹⁹Sn impurity atoms are consistent with the results of X-ray diffraction studies of *c*-GeSbTe [3,4] crystalline compounds: bivalent tin Sn²⁺ (electronic configuration $5s^2p^x$) replaces divalent germanium Ge²⁺ (electronic configuration $4s^2p^x$) in positions 4*b* crystal lattice like NaCl. The isomeric shifts of the spectra of these compounds are close to the isomeric shift of the SnTe compound, which also has a crystal lattice of the NaCl type. The spectra broadening of ternary compounds (compared to the spectral width of SnTe G = 0.94 mm/s) is due to the presence of a high concentration of stoichiometric vacancies [3,4] in the cationic sublattice of these compounds.

Thus, the germanium atoms in the structure of the *a*-GeSbTe films are tetravalent, form a tetrahedral system of chemical bonds (the local coordination number of germanium atoms is four), and tellurium atoms are predominantly located in the local environment of germanium atoms. The Messbauer spectroscopy data on ¹¹⁹Sn impurity atoms for *c*-GeSbTe films correspond to divalent tin replacing divalent germanium in the crystal lattice of the NaCl type.

Conflict of interest

The authors declare that they have no conflict of interest.

References

- D. Lencer, M. Salinga, M. Wuttig, Adv. Mater., 23, 2030 (2011). DOI: 10.1002/adma.201004255
- [2] C. Qiao, Y.R. Guo, J.J. Wang, H. Shen, S.Y. Wang, Y.X. Zheng, R.J. Zhang, L.Y. Chen, C.Z. Wang, K.M. Ho, J. Alloys Compd., 774, 748 (2019). DOI: 10.1063/5.0067157
- [3] B. Zhang, X.P. Wang, Z.J. Shen, X.B. Li, C.S. Wang, Y.J. Chen, J.X. Li, J.X. Zhang, Z. Zhang, S.B. Zhang, X.D. Han, Sci. Rep., 6, 25453 (2016). DOI: 10.1038/srep25453
- [4] X.-P. Wang, X.-B. Li, N.-K. Chen, Q.-D. Chen, X.-D. Han, S. Zhang, H.-B. Sun, Acta Mater., **136**, 242 (2017). DOI: 10.1016/j.actamat.2017.07.006
- [5] A. Lotnyk, U. Ross, S. Bernütz, E. Thelander, B. Rauschenbach, Sci. Rep., 6, 26724 (2016). DOI: 10.1038/srep26724
- [6] Y. Zheng, Y. Wang, T. Xin, Y. Cheng, R. Huang, P. Liu, M. Luo, Z. Zhang, Z. Song, S. Feng, Commun. Chem., 2, 1 (2019). DOI: 10.1038/s42004-019-0114-7
- [7] A.V. Kolobov, P. Fons, A.I. Frenkel, A.L. Ankudinov, J. Tominga, T. Uruga, Nature Mater., 3, 703 (2004). DOI: 10.1038/nmat1215
- [8] D.A. Baker, M.A. Paesler, G. Lucovsky, S.C. Agarwal, P.C. Taylor, Phys. Rev. Lett., 96, 255501 (2006).
 DOI: 10.1103/PhysRevLett.96.255501
- [9] D.A. Baker, M.A. Paesler, G. Lucovsky, P.C. Taylor, J. Non-Cryst. Solids, 352, 1621 (2006).
 DOI: 10.1016/j.jnoncrysol.2005.11.079
- P. Jóvári, I. Kaban, J. Steiner, B. Beuneu, A. Schöps, M.A. Webb, Phys. Rev. B, 77, 035202 (2008).
 DOI: 10.1103/PhysRevB.77.035202
- [11] A.V. Marchenko, E.I. Terukov, F.S. Nasredinov, Yu.A. Petrushin, P.P. Seregin, Semiconductors, 55, 1 (2021). DOI: 10.1134/S1063782621010127.
- [12] A.V. Marchenko, P.P. Seregin, E.I. Terukov, K.B. Shakhovich, Semiconductors, 53, 711 (2019).
 DOI: 10.1134/S1063782619050166.
- [13] H. Micklitz, P.H. Barrett, Phys. Rev. B, 5, 1704 (1972).
 DOI: 10.1103/PhysRevB.5.1704