Growth of thick $\varepsilon(\kappa)$ -Ga₂O₃ films by halide vapor phase epitaxy

© S.I. Stepanov, A.I. Pechnikov, M.P. Scheglov, A.V. Chikiryaka, V.I. Nikolaev

loffe Institute, St. Petersburg, Russia E-mail: s.i.stepanov@gmail.com

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High crystalline quality epitaxial films of orthorhombic gallium oxide $\varepsilon(\kappa)$ -Ga₂O₃ with a thickness of more 20 μ m have been grown for the first time using halide vapor phase epitaxy. Sapphire wafers with preliminarily deposited GaN layers were used as substrates. The properties of the produced films are studied by X-ray diffraction and electron microscopy. The results are considered an important step towards obtaining thick layers and quasi-bulk $\varepsilon(\kappa)$ -Ga₂O₃ crystals for practical applications in electronics and sensor technology.

Keywords: gallium oxide, halide vapor phase epitaxy, polymorph, X-ray diffraction

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Gallium oxide Ga₂O₃ has recently attracted research attention as a promising ultrawide-gap semiconductor material. Its energy gap ($E_g = 4.7-5.3 \text{ eV}$) Ga₂O₃ is wider than the one of common wide-gap semiconductors, such as GaN (3.4 eV) and SiC (3.0 eV). Gallium oxide may find application in deep-UV optoelectronic devices, solarblind radiation detectors, and high-power and high-voltage instruments.

Gallium oxide forms several polymorphic modifications $(\alpha, \beta, \gamma, \gamma, \delta, \alpha, \alpha, \varepsilon(\kappa))$ -phases). The most thermodynamically stable of them is β -Ga₂O₃ with a monoclinic crystal lattice. The majority of studies are focused on β -Ga₂O₃, since this polymorphic phase may be obtained in the form of both bulk single crystals and thin epitaxial films. However, the crystal structure of β -Ga₂O₃ is characterized by a low degree of symmetry, which translates into a well-pronounced anisotropy of physical properties. Single crystals and β -Ga₂O₃ epitaxial films tend to crack along the cleavage planes, creating significant challenges in device fabrication.

The interest in other polymorphic phases of Ga₂O₃ (especially in metastable α - and ε -Ga₂O₃ phases, which can be obtained by epitaxial growth on oriented substrates) has been on the rise in recent years. Since these polymorphic phases feature a crystal structure with a higher degree of symmetry and a less pronounced anisotropy of physical properties, they are better suited for device applications than β -Ga₂O₃. In contrast to other polymorphic modifications, ε -Ga₂O₃ is a ferroelectric with a relatively strong spontaneous polarization [1,2]. These properties of ε -Ga₂O₃ can be used to fabricate heterostructures with a two-dimensional electron gas and high-power switching devices with high mobility [3].

Phase-pure ε -Ga₂O₃ thin films have been produced for the first time in 2015 by halide vapor phase epitaxy (HVPE) on GaN, AlN, and β -Ga₂O₃ (201) substrates [4]. In later studies, ε -Ga₂O₃ films have also been synthesized by metal organic chemical vapor deposition (MOCVD) [5], mist CVD [6], molecular beam epitaxy [7], and pulsed laser deposition [8]. The growth of ε -Ga₂O₃ on various crystal substrates, such as GaN, AlN, SiC, ZnO, α -Al₂O₃, β -Ga₂O₃, MgO, yttria-stabilized zirconia, etc., has been reported. It should be noted that substrates with a hexagonal crystal structure (e.g., sapphire substrates with a thin GaN or AlN layer) are the best for stabilizing the ε phase.

It has been assumed in early studies [9] that ε -Ga₂O₃ features a hexagonal crystal structure with the space group $P6_3mc$. However, a detailed examination of the microstructure with a high-resolution transmission electron microscopy revealed that ε -Ga₂O₃ is not a standalone phase, but rather a combination of nanosized domains with an orthorhombic crystal structure similar to κ -Al₂O₃ with the space symmetry group $Pna2_1$ [10]. Depending on the domain size and the spatial resolution of the characterization technique, ε -Ga₂O₃ can exhibit either hexagonal (ε) or orthorhombic (κ) symmetry [11]. Therefore, a compound designation ($\varepsilon(\kappa)$ -Ga₂O₃) is used hereinbelow.

The halide vapor phase epitaxy technique is characterized by high growth rates (several tens of micrometers per hour [4,12]), which make it most suited for the fabrication of thick layers and quasi-bulk crystals of metastable polymorphic Ga_2O_3 phases. In addition, the presence of chlorine appears to be essential for the stabilization of metastable polymorphic Ga_2O_3 modifications [13].

We have already reported on the fabrication of $\varepsilon(\kappa)$ -Ga₂O₃ layers with a thickness up to 5 μ m on 2*H*-GaN, 4*H*-SiC, and smooth and patterned sapphire substrates [14]. In the present study, the thickness of fabricated $\varepsilon(\kappa)$ -Ga₂O₃ layers was increased to more than 20 μ m.

Films of Ga₂O₃ were grown by halide vapor phase epitaxy in a horizontal quartz reactor with hot walls. Patterned (0001) sapphire wafers with a predeposited gallium nitride (GaN) buffer layer were used as substrates. This layer was synthesized by MOCVD and had a thickness of $\sim 3-4\,\mu\text{m}$. Complete leveling off of protrusions on surface of the patterned sapphire substrate was thus achieved, and



Figure 1. Cross-sectional SEM image of cleaved Ga_2O_3 film on a patterned sapphire substrate with a GaN underlayer.

a planar GaN layer was formed. Gaseous oxygen (O_2) and gallium chloride (GaCl) were used as oxygen and gallium precursors. Gallium chloride vapor was produced in the reactor by flowing gaseous hydrogen chloride (HCl, 99.999%) above a boat with metallic gallium (99.9999%). The flow rate of HCl through the Ga source and the oxygen flow rate were 100 and 300 sccm, respectively. Highpurity argon was used as a carrier gas. The deposition of Ga₂O₃ films was performed at in the temperature range of 530–630°C under atmospheric pressure. The crystal structure and the crystalline quality of Ga₂O₃ films were characterized by X-ray diffraction (XRD) with CuK_{a1} $(\lambda = 1.5405 \text{ Å})$ radiation. The thickness of Ga₂O₃ films was measured by cross-sectional optical and scanning electron microscopy.

Scanning electron microscopy (SEM) images revealed that the deposited $\varepsilon(\kappa)$ -Ga₂O₃ films are uniform and have a planar surface (see Fig. 1). The thickness of films measured on the cleaved samples samples was $1-20 \,\mu$ m.

Figure 2 shows the typical XRD pattern of a synthesized Ga₂O₃ film. The diffraction peaks at $2\theta = 19.1$, 38.8, 59.8, and 83.2° can be identified as reflections (002), (004), (006), and (008) from $\varepsilon(\kappa)$ -Ga₂O₃. Intense peaks from the sapphire substrate and the GaN underlayer are also seen. The weak diffraction peak at $2\theta = 40.9^{\circ}$ can be identified as reflection (006) of α -Ga₂O₃. Since no diffraction peaks were observed at other positions, it is fair to assume that the films consisted primarily of a pure $\varepsilon(\kappa)$ -Ga₂O₃ phase (with an insignificant addition of α -Ga₂O₃).

The crystalline quality of $\varepsilon(\kappa)$ -Ga₂O₃ films was determined by measuring the full width at half maximum (FWHM) of X-ray rocking curves for the (006) reflection. Compared to other polymorphic modifications (e.g., α - and β -Ga₂O₃), $\varepsilon(\kappa)$ -Ga₂O₃ films feature a lower degree of structural perfection and are characterized by significantly higher FWHM values of rocking curves. In the majority of studies, the FWHM of X-ray rocking curves is greater than 10 arcmin [4,5,12,15], which corresponds to a density of threading dislocations on the order of 10¹⁰ cm⁻². The FWHM of the X-ray rocking curve of the (006) reflection for the $\varepsilon(\kappa)$ -Ga₂O₃ films obtained in the present study varied from 4 to 20 arcmin. Figure 3 presents the dependence of the X-ray rocking curve FWHM for the (006) reflection on the $\varepsilon(\kappa)$ -Ga₂O₃ layer thickness. It can be seen that the



Figure 2. X-ray diffraction pattern of a Ga_2O_3 layer on a patterned sapphire substrate with a GaN underlayer.



Figure 3. Dependence of the X-ray rocking curve FWHM for the (006) reflection of $\varepsilon(\kappa)$ -Ga₂O₃ on the Ga₂O₃ layer thickness.

FWHM value increases with layer thickness up to $\sim 5 \,\mu$ m; however, as the thickness grows further, the rocking curves get more narrow (i.e., the degree of crystalline perfection of $\varepsilon(\kappa)$ -Ga₂O₃ increases). The narrowest rocking curve (FWHM = 4 arcmin) corresponded to the thickest sample with the $\varepsilon(\kappa)$ -Ga₂O₃ layer thickness of 20.5 μ m.

Thus, thick single-phase $\varepsilon(\kappa)$ -Ga₂O₃ layers with to 20.5 μ m thick were synthesized by halide vapor phase epitaxy. The full width at half maximum of the X-ray rocking curve for the (006) reflection was 4 arcmin, which is presently a record low. The observed increase in the degree of structural perfection in thicker layers is indicative of the viability of application of halide vapor phase epitaxy in synthesis of thick high-quality $\varepsilon(\kappa)$ -Ga₂O₃ layers that may be used to form heterostructures or electronic and optoelectronic instruments and sensors.

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Conflict of interest

The authors declare that they have no conflict of interest.

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