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# Luminescent properties of colloidal Ag<sub>2</sub>S quantum dots for photocatalytic applications

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The structural and optical properties of colloidal Ag<sub>2</sub>S quantum dots in various environments are investigated. With the help of transmission electron microscopy, X-ray diffraction and energy-dispersive X-ray analysis the formation of colloidal Ag<sub>2</sub>S quantum dots with an average size of 2-3 nm with a monoclinic crystal lattice, and Ag<sub>2</sub>S/SiO<sub>2</sub> core-shell systems based on them, has been established. The change in the luminescence quantum yield of quantum dots with the change of the surface environment state is shown. The decoration of TiO<sub>2</sub> nanoparticles of 10-15 nm in size with Ag<sub>2</sub>S quantum dots was performed and the influence of the structure of the interfaces of quantum dots and their environment (2-mercaptopropionic acid, water, ethylene glycol, SiO<sub>2</sub> dielectric shell with a thickness of 0.6 nm and 2.0 nm) on the formation of TiO<sub>2</sub>-Ag<sub>2</sub>S heterosystems was analyzed. For Ag<sub>2</sub>S quantum dots passivated with 2-mercaptopropionic acid, signs of charge phototransfer after adsorption on TiO<sub>2</sub> nanoparticles surface have been established. Signs of reactive oxygen species appearance due to charge phototransfer in heterosystem are enstablished, based on methylene blue photobleaching under excitation of heterosystem outside of TiO<sub>2</sub> fundamental absorption region,

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### 1. Introduction

Today, development of photocatalytic systems for various applications is of a high interest [1-5]. For their creation the most suitable material is Titanium Dioxide  $(TiO_2)$  [6–10]. Titanium Dioxide has unique photocatalytic properties, however, the fundamental absorption edge of photocatalyst active phases of Titanium Dioxide - rutile and anatase is within the spectral region shorter than 400 nm [11,12]. Therefore, there is a problem of the Titanium Dioxide sensitization to visible radiation. Colloidal quantum dots of narrow-band semiconductor compounds have suitable spectral properties controlled by the size effect [13-15]. A promising candidate for the role of efficient photo sensitizer is silver sulfide  $(Ag_2S)$ . It is non-toxic, insoluble in water, and chemically stable. Ag<sub>2</sub>S colloidal quantum dots (QDs) have size-dependent spectrum of optical absorption, which can be shifted from the near IR region to the edge of the visible range [16-20]. Decoration of TiO<sub>2</sub> surface by Ag<sub>2</sub>S QD's can lead to increase in spectral sensitivity range of TiO<sub>2</sub> up to near infrared region [21]. In heterosystem TiO<sub>2</sub>-Ag<sub>2</sub>S, according to the literature data, in case of photo excitation Ag<sub>2</sub>S QDs may serve as donors of electrons [22,23].

Optical properties of silver sulfide are generally determined by the method of manufacture and selected passivator [24–26]. Sizes of passivator molecules also matter for implementation of the charge carriers transfer processes. This is why there is a separate complicated problem of compatibility of the  $TiO_2$ -Ag<sub>2</sub>S heterosystem components providing adsorption of QDs on the surface of  $TiO_2$  and efficient charge phototransfer. Such heterosystems are able to produce reactive oxygen species (ROS), which is required for creation of photobactericide coatings and devices for water purification and disinfection [27].

Properties of hybrid nanosystems TiO<sub>2</sub>-Ag<sub>2</sub>S are incompletely covered in literature.

regard In this only certain studies are known [21-23,27,28]. Because of non-stoichiometry, significant contribution into photoprocesses in Ag<sub>2</sub>S is provided by radiationless processes [29,30], which reduce both quantum yield of the QD luminescence, and the efficiency of sensitization of a photocatalyst. Therefore, selection of the conditions of synthesis of Ag<sub>2</sub>S QDs with considerable quantum output of luminescence and the structure of interfaces ensuring efficient adsorption at nanocrystals TiO<sub>2</sub> is an actual task.

This work presents the results of study of structural and optical properties of colloidal  $Ag_2S$  QDs, passivated by 2-mercaptopropionic acid, as well as TiO<sub>2</sub> nanocrystals decorated with studied QDs. The presented results demonstrate signs of efficient production of reactive oxygen species when the system is exposed to the visible range light.

### 2. Studied samples

Studied samples of colloidal Ag<sub>2</sub>S QDs were prepared by using the methodologies of colloidal synthesis realized with application of molecules of 2-mercaptopropionic acid



Figure 1. TEM-images and size distribution histograms of  $Ag_2S$  QDs samples:  $Ag_2S/2MPA/W - 1$ ,  $Ag_2S/MPTMS - 2$ ,  $Ag_2S/SiO_2 - 3$ .

(2MPA) as a passivator in water medium, and in ethylene glycol.

Used reagents: silver nitrate (AgNO<sub>3</sub>), 2-mercaptopropionic acid (2MPA), 3-mercaptopropyltrimethoxysilane (MPTMS), sodium metasilicate (Na<sub>2</sub>SiO<sub>3</sub>), ethylene glycol and sodium hydroxide (NaOH) by Sigma-Aldrich. The samples of photocatalyst of Titanium Dioxide IK-12-32 of the grade "A" were provided by Boreskov Institute of Catalysis SB RAS.

During synthesis of  $Ag_2S$  QDs in water (hereinafter referred to as  $Ag_2S/2MPA/W$ ) as a precursor we used mixture 2MPA water solution with water solution of AgNO<sub>3</sub> with the ratio [2MPA]: [AgNO<sub>3</sub>] = 1 : 2. After that, sulfur precursor was injected (Na<sub>2</sub>S) with the ratio [AgNO<sub>3</sub>] : [2MPA] : [Na<sub>2</sub>S] = 1 : 2 : 0.16 [25].

Synthesis of QP  $Ag_2S$  in ethylene glycol (hereinafter referred to as  $Ag_2S/2MPA/EG$ ) was performed according to the methodology described by us in [31].

For assembly of hybrid nanosystems  $[TiO_2-QP Ag_2S]$  nanocrytalline powder  $TiO_2$  was dissolved in water and treated luminescence quantum yield ultrasound luminescence quantum yield the 60 kHz frequency until homogeneous suspension is formed. Then, the suspension was mixed with the solution of Ag\_2S QDs and dried at room temperature. The produced dim-gray powder was washed to remove free QDs.

As reference samples featuring higher quantum yield of luminescence and low efficiency of injection of the charge carriers, we used SiO<sub>2</sub> dielectric-coated Ag<sub>2</sub>S/SiO<sub>2</sub>/2MPA QDs. The coating was formed by the method similar to that we applied in [30]. The use of MPTMS ensured the formation of monolayer SiO<sub>2</sub> shells (hereinafter referred to as Ag<sub>2</sub>S/MPTMS QDs), and an increase in the shell thickness was achieved by using an aqueous solution of Na<sub>2</sub>SiO<sub>3</sub> (hereinafter referred to as Ag<sub>2</sub>S/SiO<sub>2</sub> QDs).

# 3. Methodologies of experimental studies

For measurement of optical absorption and luminescence spectra within the region of 200–1000 nm we used spec-

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trophotometer USB-2000+XR1 (Ocean Optics, USA) with the radiation source USB-DT. For measurement of spectra of diffusion reflection spectrophotometer is additionally equipped with xenon lamp and integrating sphere. As the white standard we used barium sulfate applied to the substrate  $Al_2O_3$ .

For measurement of spectra of luminescence within the region of 800–1200 nm we used automatic spectrometer complex based on the diffraction monochromator MDR-4 (LOMO, Russia) using photo diode PDF-10C/M (ThorLabs, USA) as detector. Excitation sources were the LED module HPL-H77GV1BT-V1 (High Power Lighting Corp., Taiwan) with the wavelength of 365 nm and the laser module Osram LD PL-TB450 (Osram, Germany) with the wave length 445 nm.

Structural studies of source samples were performed by using the X-ray diffraction method by means of the diffractometer THERMO ARL X'TRA (ThermoFisher, Switzerland) and transmission electron microscopy by means of the microscope LIBRA 120 (CarlZeiss, Germany).

Measurement of luminescence quantum yield of the samples was performed by standard method of comparison with the reference [32], which was represented by the solution of indocyanine green in dymethyl sulfoxide with quantum yield of 13% [33].

# 4. Structural properties of the studied samples

Fig. 1 shows TEM images and size distribution histograms of the studied samples of Ag<sub>2</sub>S QDs.

Original ensembles of  $Ag_2S/2MPA$  QDs consists of nanocrystals with the average size of 2.2 nm and size dispersion about 40%. Analysis of high-resolution TEM image shown diffraction of electrons from crystallographic plain (122) of monoclinic lattice of  $Ag_2S$ . Addition of 0.1 ppm of MPTMS resulted in increase of the average size of crystals to 2.6 nm with dispersion of about 35%. At the same time low-contrast phase formation is observed around crystals, which can be interpreted as weak scattering of



**Figure 2.** Spectra of EDX (*a*) and X-ray diffractograms (*b*) of the studied QD samples:  $Ag_2S/2MPA/W - I$ ,  $Ag_2S/MPTMS - 2$ ,  $Ag_2S/SiO_2 - 3$ ,  $Ag_2S/2MPA/EG - 4$ .

electrons on amorphous SiO<sub>2</sub> shell. After adding sodium metasilicate there was an increase in average size of QDs up to 3.5-4.0 nm with dispersion by size about 40% and formation of thicker shell around the samples. According to high-resolution TEM-image there is low-contrast shell with the thickness of 1 nm around the crystalline core of Ag<sub>2</sub>S. Ag<sub>2</sub>S/2MPA/EG QDs have average size about 2.4 nm with the size dispersion about 30%.

Results of EDX analysis of the QD samples are given in Fig. 2.

In case of Ag<sub>2</sub>S/2MPA QDs there are emission lines of silver and sulfur atoms, as well as intensive peak of carbon atoms contained in the 2MPA molecules, as well as carbon substrate used during making the images. Presence of the lines of oxygen atoms is related to their presence in the 2MPA molecule. In case of the sample Ag<sub>2</sub>S/MPTMS the spectrum shows a low peak in the region of 1.76 keV. It is related with the group of emission lines of the silicium atoms. This line is superposition of the lines  $K_{\alpha 1}$  (1.739 keV),  $K_{\alpha 2}$  (1.739 keV) and  $K_{\beta 1}$  (1.835 keV). Also increase of intensity of the emission line  $K_{\alpha 1}$  of oxygen (0.525 keV) is observed, which is the consequence of increase of the oxygen concentration during generation of SiO<sub>2</sub> shell.

Increase of intensity of the sulfur and carbon lines occurs due to their presence in the precursor MPTMS and, probably, it is a result of completion of growing of the crystalline nucleus of QDs. For the sample Ag<sub>2</sub>S/SiO<sub>2</sub> intensity of the lines of silicium and oxygen increased, which is related with increase of the SiO<sub>2</sub> shell thickness, confirmed by the data of TEM. Also, in that sample there is a weak peak representing superposition of the lines of sodium  $K_{\alpha 1}$  (1.05 keV) and  $K_{\beta 1}$  (1.041 keV), associated with the use of precursor sodium metasilicate.

For confirmation of the crystalline structure Ag<sub>2</sub>S QDs we obtained X-ray diffractograms for study of  $K_{\alpha 1}$  copper (1.054 Å). For Ag<sub>2</sub>S/2MPA/W QDs (Fig. 2, *b*, the curve *I*), Ag<sub>2</sub>S/2MPA/EG QDs (Fig. 2, *b*, the curve *4*) in all cases on diffractograms there was a wide halo in the region 10–50°, caused by small sizes of studied crystallites. For Ag<sub>2</sub>S/2MPA/EG QDs there were low intensive peaks on the halo background, whose position corresponds to that of the reflexes of monoclinic phase of Ag<sub>2</sub>S (spatial group  $P2_{1/c}$ ) [34]. Estimations of the medium size of crystallites made by using the Scherrer formula show the presence of crystallites with the average size of ~ 3 nm. In case of Ag<sub>2</sub>S/2MPA/W QDs reflexes were not observed, which is caused by small size of crystallites.



**Figure 3.** TEM-images of the studied samples:  $TiO_2 - I$ ,  $TiO_2-Ag_2S/2MPA/W - 2$ ,  $TiO_2-Ag_2S/SiO_2 - 3$ ,  $TiO_2-Ag_2S/2MPA/EG - 4$ . Inset includes X-ray diffractogram of the sample of nanoparticles  $TiO_2$ .

As a result of replacement of passivator for  $Ag_2S/2MPA/W$  QDs from 2-MPA to MPTMS (Fig. 2, *b*, the curve 2) and further increase of the average size of crystallites there were reflexes on X-ray patterns, whose position corresponds to that of the reflexes of monoclinic phase of  $Ag_2S$  ( $P2_{1/c}$ ). Formation of additional halo in the region of angles  $15-25^{\circ}$  is associated with formation of amorphous phase  $SiO_2$  [35,36]. Further increase of the SiO<sub>2</sub> hell thickness with addition of the solution of sodium metasilicate (Fig. 1, *b*, the curve 3) resulted in increase of halo intensity in the region  $15-25^{\circ}$ . Whereas the structure of reflexes associated with monoclinic phase of  $Ag_2S$  has not changed.

Fig. 3 shows TEM-images of nanoparticles  $\rm TiO_2$  and mixtures with Ag\_S QDs.

According to TEM, the samples of TiO<sub>2</sub> represent nanoparticles with the average size of 10-15 nm (Fig. 3, *I*). There is a series of clear peaks on the X-ray pattern, that correspond to the X-ray diffraction of crystalline TiO<sub>2</sub> in the anatase phase [37]. Estimation of sizes by using the Scherrer formula showed the average size of 11-13 nm, which correlates with the TEM results. TEM images showed congestion of Ag<sub>2</sub>S and Ag<sub>2</sub>S/SiO<sub>2</sub> QDs near TiO<sub>2</sub> nanoparticles surface.

Inversely thereto, in the mixture  $TiO_2$ -Ag<sub>2</sub>S/2MPA/EG there was separation of QDs with nanoparticles. Probably,



Figure 4. Absorption (*a*) and luminescence spectra (*b*) of samples:  $Ag_2S/2MPA/W - I$ ,  $TiO_2-Ag_2S/2MPA/W - 1.1$ ,  $Ag_2S/MPTMS - 2$ ,  $Ag_2S/SiO_2 - 3$ ,  $Ag_2S/2MPA/EG - 4$ .

this is caused by behavior of ethylene glycol matrix forming micelles with QDs inside, as observed in TEM-images. Such micelles impede contact of QDs with the surface of  $TiO_2$  nanoparticles and adsorption on their surface.

# 5. Absorption properties of the studied samples

In the optical absorption spectra of the studied samples of QDs there is size effect observed. Wide bands of optical absorption specific for semiconductor QDs are shifted into short-wave region relative to the edge of fundamendal absorption of Ag<sub>2</sub>S monocrystals (1.0 eV [38]) and have the feature associated with the most probable exciton transition in the optical absorption (Fig. 4, *a*). Weak manifestation of that feature is most likely caused by heterogeneous broadening associated with QDs dispersion by size within the ensemble.

By using the data of main exciton transition position, the size of nanocrystals was estimated by using the Kayanuma Y. equation [39]. It turned out that the values of QDs average sizes lie in the range of 2.1-2.2 nm.

For the samples of Ag<sub>2</sub>S/2MPA/W and Ag<sub>2</sub>S/2MPA/EG QDs (Fig. 4, *a* the curves *1*, *4*) feature in optical absorbtion spectra corresponding to the most probable exciton transition are located at 628 and 680 nm accordingly. The results of estimations of average sizes by using the Kayanuma Y. equation were close to the data obtained by using TEM (2.2 and 2.4 nm). For the samples of Ag<sub>2</sub>S/MPTMS and Ag<sub>2</sub>S/SiO<sub>2</sub> QDs (Fig. 4, *a* the curves 2 and 3) there there is a red shift of feature in optical absorption spectra to the region to 725 nm, which is caused by growing of the

crystalline nucleus  $Ag_2S$  due to the use of sulfur-containing precursor MPTMS. Such assumption is confirmed by the data of TEM and XRD.

The most important property of  $Ag_2S$  QDs in terms of photocatalysis is the possibility of their photoexcitation within a wide range from 750 to 400 nm, which is required for spectral sensitization of anatase nanocrystals.

Fig. 5 shows spectra of optical absorption of  $TiO_2$  nanoparticles, pure, and decorated with various samples of  $Ag_2S$  QDs, obtained by the method of diffusion reflectance. The measured spectra of diffusion reflectance were built as Kubelka–Munk function:

$$F(R) = \frac{(1-R)^2}{2R},$$
 (1)

where R — diffusion reflectance.

Approximation of the linear part of function enabled determination of the edge of fundamental absorption of the studied samples of TiO<sub>2</sub>, which was about 3.2 eV, which corresponds to the bandgap width of crystalline anatase phase [40,41]. According to the obtained spectra it can be seen that TiO<sub>2</sub> (Fig. 4 the curve *I*) has absorption from 390 nm, which correlates with the data on the bandgap width of anatase (3.2 eV [40,41]). Samples of nanoparticles TiO<sub>2</sub>, decorated with Ag<sub>2</sub>S QDs, have considerable absorption within the visible region.

# 6. Luminescent properties of the samples

Fig. 4, b shows luminescence spectra of the studied samples in case of excitation by radiation with the wavelength



**Figure 5.** Spectra of optical absorption obtained by using the methodology of diffusion reflectance and Kubelka–Munk equation for the studied samples nanoparticles  $TiO_2 - I$ ; nanoparticles  $TiO_2$  decorated with QDs samples:  $Ag_2S/2MPA - 2$ ,  $Ag_2S/MPTMS - 3$ ,  $Ag_2S/SiO_2 - 4$ .



**Figure 6.** Dependence of normalized absorption within the band of methylene blue 665 nm on the time of exposure of the studied samples:  $MB^+ - I$ ,  $MB^+ + TiO_2 - 2$ ,  $MB^+ + TiO_2-Ag_2S/2MPA/W - 3$ ,  $MB^+ + TiO_2-Ag_2S/MPTMS - 4$ ,  $MB^+ + TiO_2-Ag_2S/2MPA/W - 5$ ,  $MB^+ + TiO_2-Ag_2S/2MPA/EG - 6$ ,  $MB^+ + Ag_2S/2MPA/W - 7$ ,  $MB^+ + Ag_2S/MPTMS - 8$ ,  $MB^+ + Ag_2S/SiO_2 - 9$ ,  $MB^+ + Ag_2S/2MPA/EG - 10$ .

of 445 nm. Ag<sub>2</sub>S/2MPA/W QDs (Fig. 4, *b* the curve *I*) demonstrate luminescence in the near IR-region with the maximum in the region of 850 nm and quantum yield of 0.17%. The large full width at half maximum (FWHM) of the luminescence band and considerable Stokes shift relative to exciton transition in absorption, indicate luminescence

occurring as a result of recombination of free hole with localized electron [24].

Replacement of solvent with ethylene glycol in case of Ag<sub>2</sub>S/2MPA/EG QDs (Fig. 4, *b*, the curve 4) results in shift of the maximum of luminescent band to 810 nm and increase of its quantum yield to 1.8%. The observed changes are caused by the interface nature of the centers of luminescence, which was discussed in publications many times [24–26,31].

Formation of shells results in 2.8 times increase of the quantum output of QDs luminescence up to 0.42 and 0.47%, accordingly. There is also short-wave shift of the maximum of luminescence by 45 nm with decrease of the Stokes shift relative to exciton transition in the absorption from 222 to 100 nm. Change in spectrum of luminescence is caused by formation of SiO<sub>2</sub> shell and change of interfaces of Ag<sub>2</sub>S [42] QDs, as well as by limitation of penetration of the charge carriers into environment. The observed dependences turned out to be similar to the case when formation of Ag<sub>2</sub>S/SiO<sub>2</sub> QDs takes place in ethylene glycol [30].

Decoration of TiO<sub>2</sub> nanoparticles with Ag<sub>2</sub>S QDs resulted in changes in the luminescence spectrum only in case of the sample  $Ag_2S/2MPA/W$  QDs (Fig. 4, b, the curves 1 and 1.1). For this sample the 25 times decrease in the luminescence quantum yield from 0.17 to 0.007% were observed. Taking into account the data of TEM, which demonstrate agglomeration of QDs near to the surface of Titanium Dioxide nanoparticles, the conclusion about possible charge carriers phototransfer were made. For other samples intensity and spectra of luminescence had not changed. In each of three cases QDs featured the shell of SiO<sub>2</sub>, or viscous ethylene glycol. Therefore, regardless of the best luminescent properties of Ag<sub>2</sub>S/2MPA/EG QDs, the indications of charge carriers phototransfer and formation of TiO<sub>2</sub>-Ag<sub>2</sub>S QDs is specific only for the case of decoration with Ag<sub>2</sub>S/2MPA/W QDs.

Such conclusion is confirmed by the results of analysis of photobleaching of the methylene blue dye [43-45] in presence of the studied heterosystems (Fig. 6). Changes of optical density of methylene blue monomer absorption band (650–660 nm) were controlled during excitation of heterosystem TiO<sub>2</sub>-Ag<sub>2</sub>S QDs with 445 nm radiation, which is beyond the region of TiO<sub>2</sub> absorption. Bleaching of methylene blue was observed only in the sample TiO<sub>2</sub>-Ag<sub>2</sub>S/2MPA/W (Fig. 6, the curve 3). For all other samples, including separate components of heterosystem, there were no changes of optical density of methylene blue.

### 7. Conclusion

In this work, the analysis of structural and optical properties of colloidal  $Ag_2S$  QDs synthesized in various environment ( $Ag_2S/2MPA$ ,  $Ag_2S/MPTMS$ ,  $Ag_2S/SiO_2$  in water,  $Ag_2S/2MPA$  in ethylene glycol) was made. The dependence of luminescence quantum yield of  $Ag_2S$  QDs on their interface state was studied. Increase of the

luminescence quantum yield of Ag<sub>2</sub>S QDs up to 2.8 times (from 0.17 to 0.47%) as a result of SiO<sub>2</sub> shell formation was obseved. Impact of QDs environment on the possibility of charge carriers transfer between the components if TiO2-Ag2S QDs heterosystem was demonstrated and possibility of TiO2-Ag2S QDs heterostructure formation was analysed. As a result of analysis it was determined that the observed reduction of luminescence quantum yield from 0.17 to 0.007% at decoration of TiO<sub>2</sub> nanoparticles with Ag<sub>2</sub>S/2MPA/W QDs is an indication of the charge phototransfer between the components of heterosystem. In other cases (Ag<sub>2</sub>S/MPTMS, Ag<sub>2</sub>S/SiO<sub>2</sub> in water, Ag<sub>2</sub>S/2MPA in ethylene glycol) no such spectral manifestations were found. Formation of heterosystems TiO2-Ag2S enables producing the reactive oxygen species during excitation with visible range radiation, which was demonstrated by the photobleaching of methylene blue in presence of TiO2 nanoparticles, decorated with Ag<sub>2</sub>S/2MPA/W QDs. Use of mixtures of TiO<sub>2</sub> nanoparticles with water solutions of Ag<sub>2</sub>S/MPTMS and Ag<sub>2</sub>S/SiO<sub>2</sub> QDs, as well as Ag<sub>2</sub>S/2MPA QDs in ethylene glycol did not provide sensitization of TiO<sub>2</sub> nanoparticles at that spectral range. Based on the obtained data we concluded that the possibility of formation of heterosystems TiO<sub>2</sub>-Ag<sub>2</sub>S, able of producing reactive oxygen species during excitation with visible range radiation, is highly dependent on the Ag<sub>2</sub>S QDssurrounding, which ensures proper passivation of its interfaces and not impedes the charge transfer between the components of heterosystem.

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#### **Conflict of interest**

The authors declare that they have no conflict of interest.

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