

07,01

## Conditions of elastic mechanical stability and elastic properties of crystal structures with different symmetry

© A.I. Gusev, S.I. Sadovnikov

Institute of Solid State Chemistry, Russian Academy of Sciences, Ural Branch, Yekaterinburg, Russia

E-mail: gusev@ihim.uran.ru

Received February 19, 2022

Revised March 2, 2022

Accepted March 3, 2022

A critical analysis of the conditions of elastic stability of crystal structures of different symmetry as restrictions imposed on their elastic constants  $c_{ij}$  has been carried out. It is shown that the conditions of elastic stability of all crystals, except for cubic ones, are described by polynomials from the second to sixth powers of their elastic constants  $c_{ij}$ . Necessary and sufficient conditions for the elastic stability of crystals of different symmetry are presented explicitly.

**Keywords:** Elastic stability, Symmetry, Elastic constants, Stability conditions.

DOI: 10.21883/PSS.2022.06.53829.292

### 1. Introduction

To study elastic properties of the crystal materials is a priority task in the solid-state physics. Along with experimental studies, the last two decades have witnessed distribution *ab initio* of calculations of elastic constants and modules of solid-phase compounds with a various structure by using the density functional theory.

The electron structure and elastic properties of simulated crystal compounds are calculated by a full-potential linearized augmented-plane-wave method including a local-orbital method (FP-LAPW + *lo*) [1,2], which are based on the density functional theory (DFT).

Recently, theoretical methods have been intensely developed to simulate possible structures of compounds and their mechanical properties depending on the composition and to find structures to be favorable in terms of energy. Quite reliable results can be obtained by means of an evolution prediction algorithm of crystal structures as implemented in the USPEX software [3–5].

The energy of the crystal compounds simulated by means of the evolution algorithm is calculated withing the density functional theory [6] using the PBE version [7] in the Generalized Gradient Approximation (GGA) and the Projector-Augmented Wave method (PAW) [8], as implemented in the VASP code (Vienna Ab initio Simulation Package) [9,10].

There are hundreds of original studies in the literature, which are devoted to evaluate the elastic properties of a variety of crystal substances.

The structure and elastic properties of osmium nitride OsN<sub>2</sub> with cubic and rhombic lattices, ultra-hard iridium nitrides IrN<sub>2</sub> and IrN<sub>3</sub> with cubic, hexagonal, tetragonal, monoclinic and several rhombic structures, platinum nitride PtN with structures of a zinc-blende type, rhombic and

cubic structures were studied in the papers [11–13]. The structure and mechanical stability of the elastic properties of niobium nitrides NbN with the NaCl and CsCl cubic structures and the hexagonal  $\delta$ - and  $\epsilon$ -structures as well as of tetragonal and cubic modifications of boron nitride are discussed in the papers [14,15].

Tetragonal, orthorhombic and monoclinic phases of silver azide AgN<sub>3</sub> are discussed in terms of their elastic properties and mechanical stability in the paper [16], which shows significant difference in their stability and usability of the orthorhombic modification of the silver azide in the solid-state reaction of explosion decomposition. The elastic properties and mechanical stability of the cubic, orthorhombic and monoclinic chlorates and perchlorates NaClO<sub>3</sub>, KClO<sub>3</sub>, LiClO<sub>4</sub>, NaClO<sub>4</sub>, KClO<sub>4</sub> are studied in the paper [17].

The structure and mechanical properties of the orthorhombic and trigonal phases CaB<sub>2</sub>H<sub>2</sub>, the orthorhombic and monoclinic polymorphs Li<sub>2</sub>FeSiO<sub>4</sub>, which can be alternative cathode materials, are discussed in the papers of Vajeeston et al. [18,19]. The *ab initio* calculations of the mechanical properties of the organic & inorganic hybrid perovskites CH<sub>3</sub>NH<sub>3</sub>BX<sub>3</sub> ( $B = \text{Sn, Pb}$ ;  $X = \text{Br, I}$ ) with cubic, tetragonal and orthorhombic structures are carried out within the density functional theory in the paper [20]. The elastic properties and mechanical stability of the semiconductor cubic solid solutions In<sub>x</sub>Al<sub>1-x</sub>As<sub>y</sub>Sb<sub>1-y</sub> are studied in the paper [21], while the paper [22] discusses the mechanical stability of gypsum CaSO<sub>4</sub> · 2H<sub>2</sub>O.

Methods of calculation of the elastic stiffness and mechanical stability constants can be applied not only to the solid-phase crystal compounds, but to fullerites and materials based thereon [23–25].

Structures of numerous carbide phases  $M_nC_m$  ( $M = \text{Zr, Hf, Nb, Ta}$ ) [26–32] were predicted and calculated

by means of theoretical methods of analysis of the electron and crystal structure.

The persistence of the crystal structure is importantly characterized by an elastic (mechanical) stability, whose criteria are defined by the elastic constants. The conditions of the mechanical stability (persistence) of the crystal structures are usually worded as requirements to the elastic stability of the crystal lattices and depend on their symmetry [33]. Criteria of the Born elastic stability are well known for cubic crystals, and the monograph [33] explicitly provides the stability criteria of cubic, hexagonal, tetragonal and trigonal crystals. However, for crystals with lower symmetry (especially, rhombic and monoclinic ones), the conditions of the mechanical persistence are defined inexactly and even erroneously (see, for example, the paper [12]). Derivation of necessary and sufficient conditions of the elastic stability of the crystals is a nontrivial and quite difficult task. That is why many authors just cite wrong stability conditions from the paper [12] and use them. In connection therewith, with no sophisticated mathematical derivation, the present paper provides necessary and sufficient conditions of the elastic stability of crystals of a various symmetry.

## 2. Results and discussion

Depending on the crystal symmetry, there are 7 qualitatively different matrices of the elastic constants, which differ in a number of independent non-zero variables  $c_{ij}$  and correspond to the seven crystal systems [33]. The paper [34] discusses 9 qualitatively different matrices (two matrices each, with a different number of non-zero variables  $c_{ij}$  for tetragonal and trigonal crystals with a different point symmetry). However, Fedorov showed in the monograph [33] that for correct selection of orientation of the coordinate system for all the seven point crystal symmetry groups of the tetragonal system there is one matrix of the elastic constants; for the trigonal crystals with five point symmetry groups there is also one matrix of the elastic constants.

The elastic behavior of the lattice is described by the matrix of the elastic constants of the second order:

$$c_{ij} = \frac{1}{V_0} \left( \frac{\partial^2 E}{\partial \varepsilon_i \partial \varepsilon_j} \right), \quad (1)$$

where  $E$  and  $V_0$  — energy of the elastic deformation of the crystal and its equilibrium volume,  $\varepsilon$  — deformation. In the general case, the matrix  $(\mathbf{C})$  of the elastic stiffness constants has a size of  $6 \times 6$ , is a symmetrical one and can include 21 independent elastic constants  $c_{ij}$ .

The energy of elastic deformation referred to a crystal volume unit, for arbitrary infinitely small deformation is determined in [33,35,36] as

$$E/V \sim \frac{1}{2} \sum_{i,j=1}^6 c_{ij} \varepsilon_i \varepsilon_j. \quad (2)$$

The matrix  $(\mathbf{C})$  of the elastic stiffness constants of low-symmetry crystals of a triclinic system includes the biggest number of the independent non-zero elastic constants — 21 ( $c_{11}, c_{12}, c_{13}, c_{14}, c_{15}, c_{16}, c_{22}, c_{23}, c_{24}, c_{25}, c_{26}, c_{33}, c_{34}, c_{35}, c_{36}, c_{44}, c_{45}, c_{46}, c_{55}, c_{56}$  and  $c_{66}$ ) and takes the following form:

$$(\mathbf{C})_{\text{tricl}} = \begin{pmatrix} c_{11} & c_{12} & c_{13} & c_{14} & c_{15} & c_{16} \\ c_{12} & c_{22} & c_{23} & c_{24} & c_{25} & c_{26} \\ c_{13} & c_{23} & c_{33} & c_{34} & c_{35} & c_{36} \\ c_{14} & c_{24} & c_{34} & c_{44} & c_{45} & c_{46} \\ c_{15} & c_{25} & c_{35} & c_{45} & c_{55} & c_{56} \\ c_{16} & c_{26} & c_{36} & c_{46} & c_{56} & c_{66} \end{pmatrix}. \quad (3)$$

The matrix  $(\mathbf{S})$  of elastic compliance constants is correlated to the matrix  $(\mathbf{C})$  in a simple relationship  $(\mathbf{S})^{-1} = (\mathbf{C})$  or  $(\mathbf{C})(\mathbf{S}) = 1$ . Taking this into account, the matrix  $(\mathbf{S})$  of the elastic compliance constants has the same size of  $6 \times 6$  and includes 21 independent elastic constants  $s_{ij}$ . For crystals with the symmetry higher than the triclinic one, a part of the constants  $c_{ij}$  or  $s_{ij}$  goes to 0.

As per the studies [35–37], the crystal is stable if and only if the energy of the elastic deformation is positive, i.e. above zero for all the actual values  $\varepsilon_i$  and  $\varepsilon_j$ , if only all  $\varepsilon_i$  and  $\varepsilon_j$  are nonzero. It imposes additional limitations on the constants  $c_{ij}$  and  $s_{ij}$ .

The mechanical stability of the crystal of an arbitrary symmetry has a necessary condition for it, which is not a sufficient one: all diagonal elements of the matrix of the elastic stiffness constants are to be positive, i.e.

$$c_{ii} > 0 \quad (i = 1-6). \quad (4)$$

The sufficient conditions of the mechanical stability are fulfilled if all the own values of the matrix  $(\mathbf{C})$  of the elastic stiffness constants are positive. In order to determine own values of a square matrix  $C$ , a characteristic matrix  $C - \lambda E$  is used, where  $E$  — a unit matrix,  $\lambda$  — some unknown variable [38,39]. A polynomial  $|C - \lambda E|$  is a characteristic polynomial of the matrix  $C$ , and its roots are characteristic roots, i.e. are own values of the matrix  $C$ .

The elastic matrix of the cubic crystals includes only 3 independent elastic constants  $c_{11}, c_{12}$  and  $c_{44}$ :

$$(\mathbf{C})_{\text{cub}} = \begin{pmatrix} c_{11} & c_{12} & c_{12} & & & \\ c_{12} & c_{11} & c_{12} & & & \\ c_{12} & c_{12} & c_{11} & & & \\ & & & c_{44} & & \\ & & & & c_{44} & \\ & & & & & c_{44} \end{pmatrix}. \quad (5)$$

For the cubic crystals, the constants  $c_{11}, c_{12}$  and  $c_{44}$  are positive. The mechanical stability conditions of the cubic crystals determined in the papers [40,41] are known as the

Born stability criteria:

$$c_{11} > c_{12}, \quad c_{44} > 0, \quad c_{11} + 2c_{12} > 0. \quad (6)$$

The cubic system is the only one, for which the mechanical stability conditions are linear.

The elastic matrix of the hexagonal crystals includes the 5 independent elastic constants  $c_{11}, c_{12}, c_{13}, c_{33}, c_{44}$ :

$$(\mathbf{C})_{\text{hex}} = \begin{pmatrix} c_{11} & c_{12} & c_{13} & & & \\ c_{12} & c_{11} & c_{13} & & & \\ c_{13} & c_{13} & c_{33} & & & \\ & & & c_{44} & & \\ & & & & c_{44} & \\ & & & & & (c_{11} - c_{12})/2 \end{pmatrix}. \quad (7)$$

The necessary and sufficient conditions of the elastic stability of the hexagonal crystals take the following form:

$$c_{11} > |c_{12}|, \quad c_{44} > 0, \quad (c_{11} + c_{12})c_{33} > 2c_{13}^2. \quad (8)$$

The necessary and sufficient conditions of the elastic stability of the low-symmetry (tetragonal, trigonal (rhombohedral), orthorhombic and monoclinic) crystals have been earlier discussed in the papers [12,34].

The matrix of the elastic stiffness constants of the tetragonal crystals depends on the 6 independent constants  $c_{11}, c_{12}, c_{13}, c_{33}, c_{44}$  and  $c_{66}$ :

$$(\mathbf{C})_{\text{tetr}} = \begin{pmatrix} c_{11} & c_{12} & c_{13} & & & \\ c_{12} & c_{11} & c_{13} & & & \\ c_{13} & c_{13} & c_{33} & & & \\ & & & c_{44} & & \\ & & & & c_{44} & \\ & & & & & c_{66} \end{pmatrix}. \quad (9)$$

The mechanical stability conditions of the tetragonal crystals take the following form:

$$c_{11} > |c_{12}|, \quad c_{44} > 0, \quad c_{66} > 0, \quad (c_{11} + c_{12})c_{33} > 2c_{13}^2. \quad (10)$$

The mechanical stability criteria (10) of the tetragonal crystals contain the quadratic polynomials from  $c_{ij}$ , but in the papers [12,13,15] the mechanical stability conditions of the tetragonal phases are represented in an erroneous linear form as

$$c_{ii} > 0, \quad (c_{11} - c_{12}) > 0, \quad (c_{11} + c_{33} - 2c_{13}) > 0, \\ (2c_{11} + 2c_{12} + c_{33} + 4c_{13}) > 0. \quad (11)$$

The trigonal (rhombohedral) crystals have 6 independent elastic constants  $c_{11}, c_{12}, c_{13}, c_{14}, c_{33}, c_{44}$ :

$$(\mathbf{C})_{\text{trig}} = \begin{pmatrix} c_{11} & c_{12} & c_{13} & c_{14} & & \\ c_{12} & c_{11} & c_{13} & -c_{14} & & \\ c_{13} & c_{13} & c_{33} & & & \\ c_{14} & -c_{14} & & c_{44} & & \\ & & & & c_{44} & c_{14} \\ & & & & c_{14} & (c_{11} - c_{12})/2 \end{pmatrix}. \quad (12)$$

Their mechanical stability conditions take the following form:

$$c_{11} > |c_{12}|, \quad c_{44} > 0, \quad (c_{11} + c_{12})c_{33} > 2c_{13}^2, \\ (c_{11} - c_{12})c_{44} > 2c_{14}^2. \quad (13)$$

The elastic stiffness matrix of the orthorhombic crystals includes 9 independent constants  $c_{11}, c_{12}, c_{13}, c_{22}, c_{23}, c_{33}, c_{44}, c_{55}$  and  $c_{66}$ :

$$(\mathbf{C})_{\text{orthorh}} = \begin{pmatrix} c_{11} & c_{12} & c_{13} & & & \\ c_{12} & c_{22} & c_{23} & & & \\ c_{13} & c_{23} & c_{33} & & & \\ & & & c_{44} & & \\ & & & & c_{55} & \\ & & & & & c_{66} \end{pmatrix}. \quad (14)$$

The paper [34] has shown that taking into account the condition  $c_{ii} > 0$  ( $i = 1-6$ ) (4), the necessary and sufficient conditions of the mechanical stability of the rhombic crystals include quadratic and cubic polynomials and take the following form:

$$\left. \begin{aligned} c_{11} > 0, \quad c_{44} > 0, \quad c_{55} > 0, \quad c_{66} > 0, \quad c_{11}c_{22} > c_{12}^2, \\ c_{11}c_{12}c_{33} + 2c_{12}c_{13}c_{23} - c_{11}c_{23}^2 - c_{22}c_{13}^2 - c_{33}c_{12}^2 > 0 \end{aligned} \right\}. \quad (15)$$

The nonlinear conditions (15) are fundamentally different from the linear conditions:

$$c_{ii} > 0, \quad c_{ii} + c_{jj} - 2c_{ij} > 0, \\ c_{11} + c_{22} + c_{33} + 2(c_{12} + c_{13} + c_{23}) > 0, \quad (16)$$

proposed by the authors of the paper [11] and repeated in the paper [12]. Note that the stability conditions of the rhombic crystals used in the papers [11,12], are, in the best case scenario, only necessary, but not sufficient. The erroneous linear conditions (16) seem to have been obtained as a result of incorrect generalization of the cubic criteria (6) to other orthogonal (tetragonal and rhombic) crystals. Evaluation of the mechanical stability of the rhombic crystals by means of the linear conditions (16) can lead to erroneous conclusions.

In particular, the erroneous linear conditions (16) have been used to evaluate the mechanical stability



- [13] S. Patil, S. Khare, B. Tuttle, J. Bording, S. Kodambaka. *Phys. Rev. B* **73**, 10, 104118 (2006).
- [14] X.F. Li, Z.L. Liu. *J. Atom. Mol. Sci.* **3**, 1, 78 (2012).
- [15] C.-Y. Niu, J.-T. Wang. *Phys. Lett. A* **378**, 30-31, 2303 (2014).
- [16] V.M. Lisitsyn, Yu.N. Zhuravlev. *Izv. Tomsk. politekhn. un-t* **317**, 2, 138 (2010) (in Russian).
- [17] D.V. Korabel'nikov, Yu. N. Zhuravlev. *Physics of the Solid State* **58**, 6, 1129 (2016) (in Russian).
- [18] P. Vajeeston, P. Ravindran, B. Hauback, H. Fjellvåg. *Int. J. Hydrogen Energy* **36**, 16, 10149 (2011).
- [19] P. Vajeeston, H. Fjellvåg. *RSC Adv.* **7**, 27, 16843 (2017).
- [20] J. Feng. *APL Mater.* **2**, 8, 081801 (2014).
- [21] H. Algarni, O.A. Al-Hagan, N. Bouarissa, T.F. Alhuwaymel, M. Ajmal Khan. *Philosoph. Mag. A* **98**, 28, 2582 (2018).
- [22] J. Fu. In: *Density Functional Calculations — Recent Progresses of Theory and Application* / Ed. G. Yang. IntechOpen, London (2018). P. 219–241.
- [23] L.Kh. Rysaeva, Yu.A. Baimova. *Fund. probl. sovr. materialoved.* **12**, 4, 439 (2015) (in Russian).
- [24] L.Kh. Rysaeva, Yu.A. Baimova, D.S. Lisovenko, K.A. Krylova, S.V. Dmitriev, V.A. Gorodtsov. *Fund. probl. sovr. materialoved.* **13**, 1, 105 (2016) (in Russian).
- [25] L.Kh. Rysaeva. *J. Phys.: Conf. Ser.* **938**, 012071 (2017).
- [26] Q. Zeng, J. Peng, A.R. Oganov, Q. Zhu, C. Xie, X. Zhang, D. Dong, L. Zhang, L. Cheng. *Phys. Rev. B* **88**, 21, 214107 (2013).
- [27] C. Xie, A.R. Oganov, D. Li, T.T. Debela, N. Liu, D. Dong, Q. Zeng. *Phys. Chem. Chem. Phys.* **18**, 17, 12299 (2016).
- [28] X.-X. Yu, C.R. Weinberger, G.B. Thompson. *Comput. Mater. Sci. A* **112**, 318 (2016).
- [29] N. Zhang, B. Liu, J. Wang Jingyang Wang. *Acta Mater.* **111**, 232 (2016).
- [30] M.G. Kostenko, A.V. Lukoyanov, A.A. Valeeva, A.I. Gusev. *J. Exp. Theor. Phys.* **129**, 5, 863 (2019).
- [31] M.G. Kostenko, Jingyu Li, Zhi Zeng, Y.-Sh. Zhang, S.V. Sharf, A.I. Gusev, A.V. Lukoyanov. *J. Alloys Comp.* **891**, 162063 (2022).
- [32] M.G. Kostenko, A.I. Gusev, A.V. Lukoyanov. *Acta Mater.* **223**, 117449 (2022).
- [33] F.I. Felorov. *Teoriya uprugikh voln v kristallakh.* Nauka, (1965). 388 p. (in Russian).
- [34] F. Mouhat, F.-X. Coudert. *Phys. Rev. B* **90**, 22, 224104 (2014).
- [35] J.F. Nye. *Physical Properties of Crystals.* Clarendon Press — Oxford Univ. Press, Oxford (1985). 329 p.
- [36] R.E. Newnham. *Properties of Materials: Anisotropy, Symmetry, Structure.* Oxford Univ. Press, Oxford, N.Y. (2005). 378 p.
- [37] G. Grimvall, B. Magyari-Köpe, V. Ozoliņš, K.A. Persson. *Rev. Mod. Phys.* **84**, 2, 945 (2012).
- [38] A.G. Kurosh. *Kurs vysshey algebrы.* Nauka, M. (1971). 482 p. (in Russian).
- [39] *Linear Algebra and Its Applications.* 3rd ed. / Ed. G. Strang. Brooks Cole, Massachusetts (1988). 520 p.
- [40] M. Born. *Math. Proc. Camb. Phil. Soc.* **36**, 2, 160 (1940).
- [41] M. Born, K. Huang. *Dynamical Theory of Crystal Lattices.* Oxford Univ. Press, Oxford, N.Y. (1998). 432 p.
- [42] E. Calderon, M. Gauthier, F. Decremps, G. Hamel, G. Syfosse, A. Polian. *J. Phys.: Cond. Matter* **19**, 5, 436228 (2007). 13 p.